

Adhesion and Thermomechanical Reliability for PV Devices and Modules

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Steve Lin (Vitex), Matthew Robinson (Nanosolar), Christine McGuinness and Darin Laird
(Plextronics).

Solar Outlook

Then...

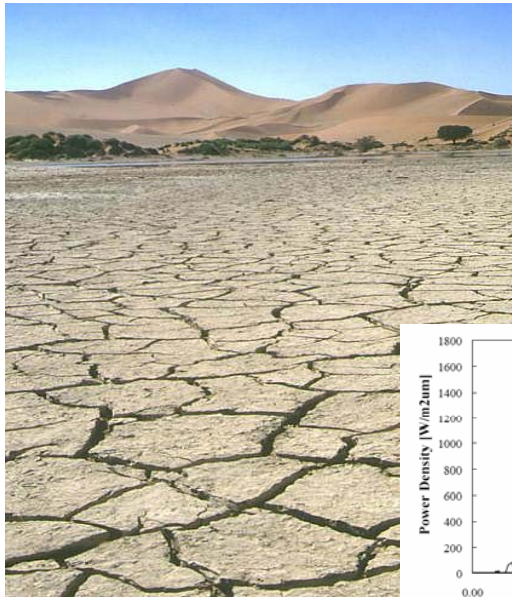
1. Efficiency
2. Reliability
3. Cost

To make a difference...

1. Cost
2. Reliability
3. Efficiency

- Engineer durable solar technologies with robust and predictable lifetimes. Start early in development – avoid roadblocks.
- Leverage from reliability physics in microelectronics – thin-film metrologies, kinetic models, accelerated tests, life prediction.
- Are degradation processes coupled and how?
- Kinetic models for damage evolution - basis for life prediction and accelerated testing (T, environment, stress, solar flux, etc.)
- Effective defensive strategies – e.g. transparent barriers with anti-reflective properties.

Degradation and Reliability of PV Devices and Modules



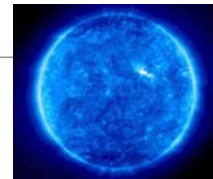
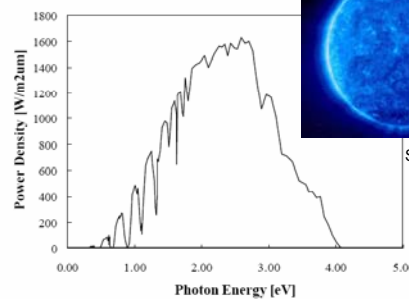
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photo from Keele University

Severe operating environments.

Exposure to thermal cycling, stress, moisture, chemically active environmental species, and UV.

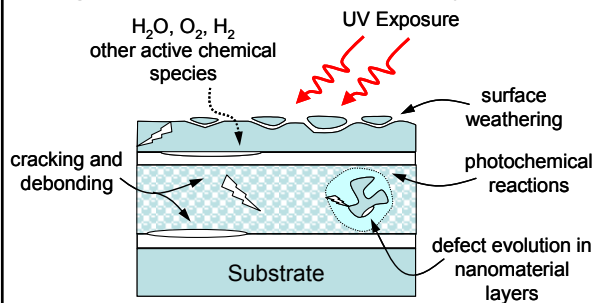
Uncertain degradation kinetics and reliability models.



Sun in UV

Solar spectrum from B. Van Zeghbroeck, U. Colorado

Degradation and Reliability of PV Devices and Modules



Severe operating environments.

Exposure to thermal cycling, stress, moisture, chemically active environmental species, and UV.

Uncertain degradation kinetics and reliability models.

Activation Energies for Bond Rupture:

VDW ~ 0.001 - .4 eV/bond

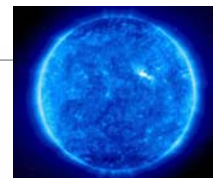
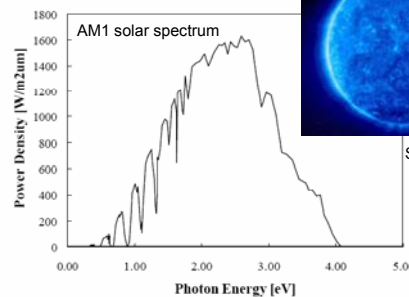
H-bonds ~ .1 - .4 eV/bond

Ester Hydrolysis (100% RH)

$E^* \sim .81$ eV/bond

SiO_2 cracking in water

$E^* \sim 1.39$ eV/bond

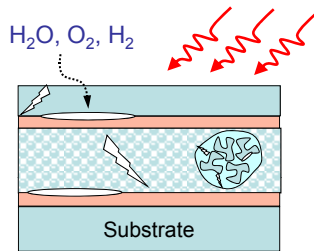


Sun in UV

Solar spectrum from B. Van Zeghbroeck, U. Colorado

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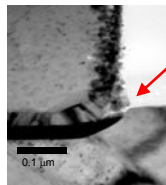
Mechanics of Damage Evolution



Damage occurs when...

$$\underbrace{G}_{\text{crack driving "force"}} \geq \underbrace{G_c}_{\text{cracking "resistance"}} \left[J / m^2 \right]$$

- Lower thin-film stresses – driving force for cracking $G = \frac{Z \sigma_f^2 h_f}{E_f}$
 - geometry/structure effects Z
 - low modulus organic films
 - multiple films, packaging, flexing
- Optimize fracture resistance G_c – cohesion of layers
 - adhesion of interfaces



nano-scale defect

Evolution of Defects and Device Reliability

absence of chemically active environmental species, damage propagates if

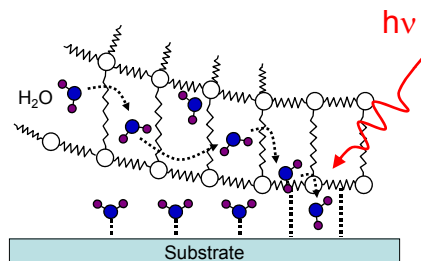
$$G \geq G_c \left[J / m^2 \right]$$

presence of chemical species and photons, damage propagates even if

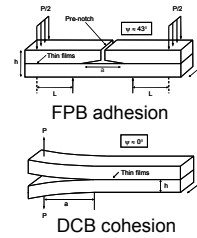
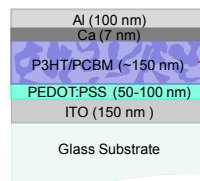
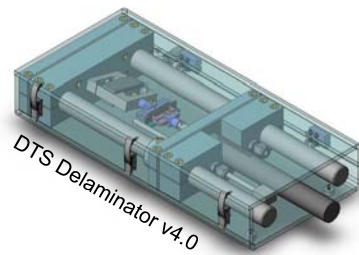
$$G < G_c \left[J / m^2 \right] \quad \text{environment and stress accelerates defect evolution}$$

Role of coupled kinetic parameters:

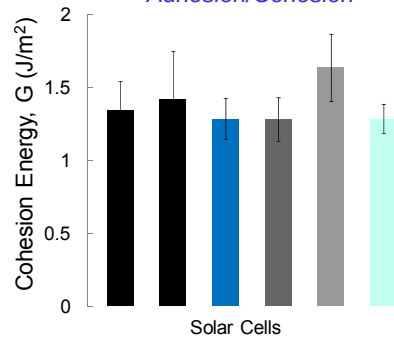
- *mechanical stress*
- *temperature*
- *environmental species*
- *photons (photochemical reactions)*



Quantitative Adhesion/Cohesion and Debond Kinetics

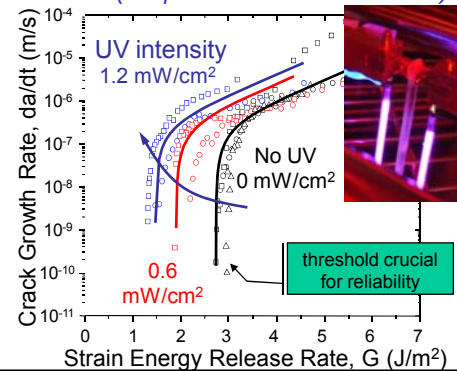


Adhesion/Cohesion



System and support dauskardt@stanford.edu

Degradation Kinetics (temp/environment /UV effects)

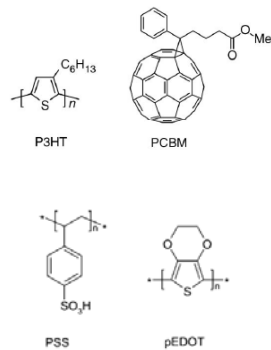


Inherent Solar Cell Thermomechanical Reliability, G_c

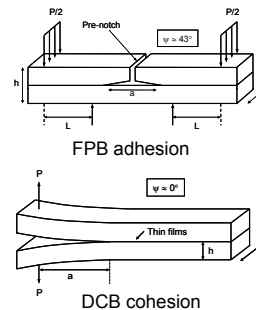
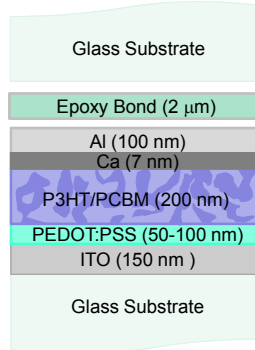
Grad students: Vitali Brand, Jeff Yang and Chris Bruner

Adhesion/Cohesion Sample Preparation

Thin films sandwiched between elastic substrates



Gunes, et. Al. Chem. Rev. 2007.

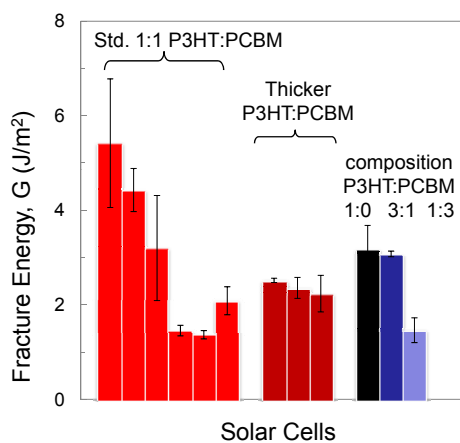


Fabricated 4-point bend adhesion and DCB cohesion test structures using standard epoxy bonding techniques.

Similar transparent glass substrates on each side.

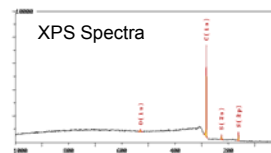
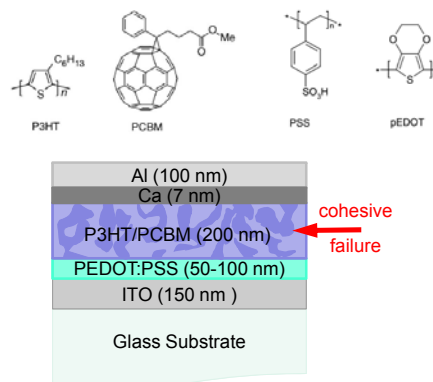


Adhesion/Cohesion of P3HT/PCBM Structures

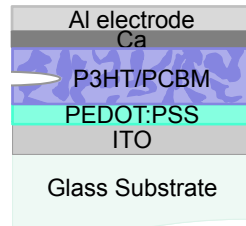
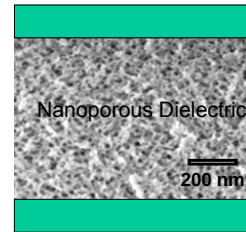
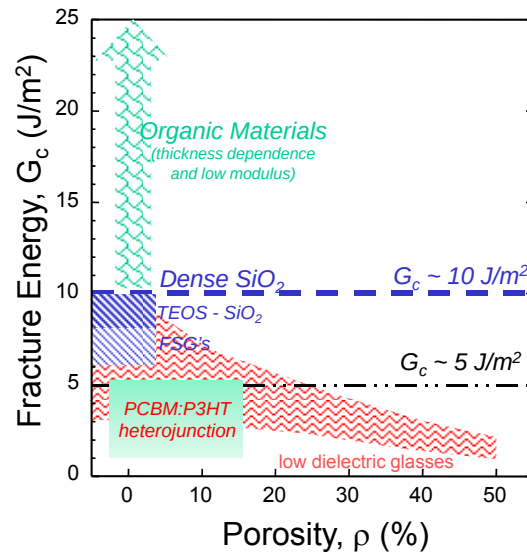


Solar Cells

- XPS reveals similar debond path for DCB and 4-pt bend samples
- C ~ 92%, S ~ 6%, O ~ 2%
- Suggests cohesive failure in PCBM:P3HT layer

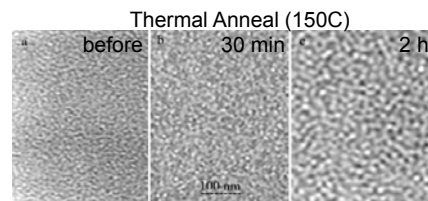
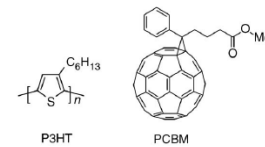
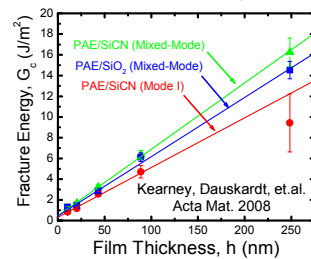


Cohesive Fracture Properties of Dielectrics



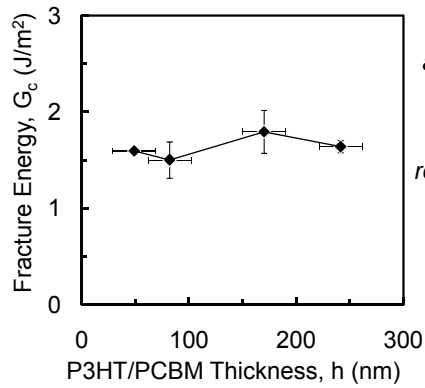
Factors Effecting Cohesion of P3HT/PCBM Layers

- Heterojunction layer thickness
 - cohesion in polymer layers is sensitive to layer thickness
 - plastic energy dissipation in organic layers
- Composition of the heterojunction layer
 - limited bonding to fullerene – expect low cohesion
 - preliminary measurements indicate higher ratios of P3HT to PCBM make stronger active layer
- Annealing
 - morphology of the P3HT:PCBM film changes with annealing, expect effect morphology on cohesion



TEM of P3HT:PCBM film
Heeger, et. Al. Adv. Func. Mat. 2005.

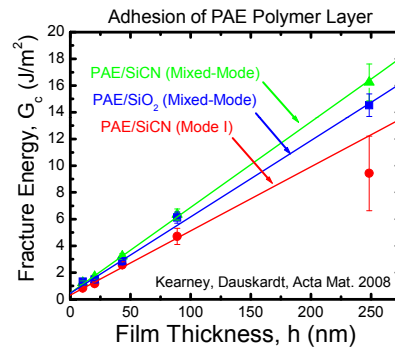
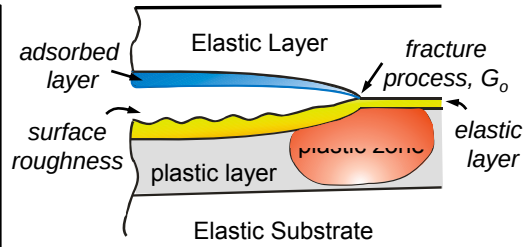
Effect of Thickness of BHJ on Cohesion



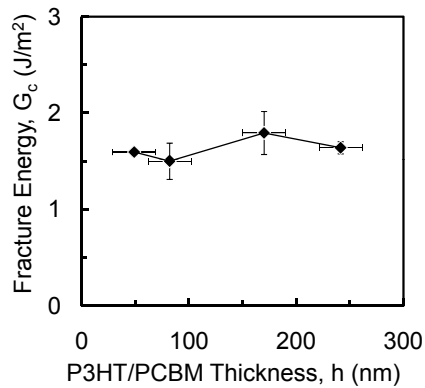
interface chemistry

$$G_{\text{total}} = G_o + G_{\text{zone}}$$

frictional contact
ductile adjacent layer



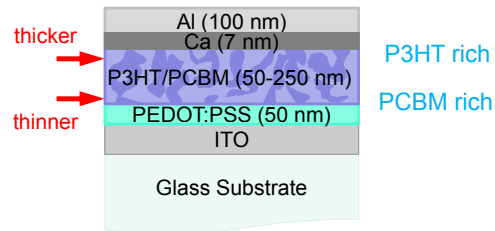
Effect of Thickness of BHJ on Cohesion



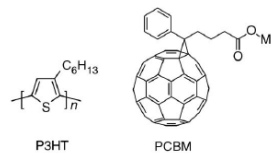
interface chemistry

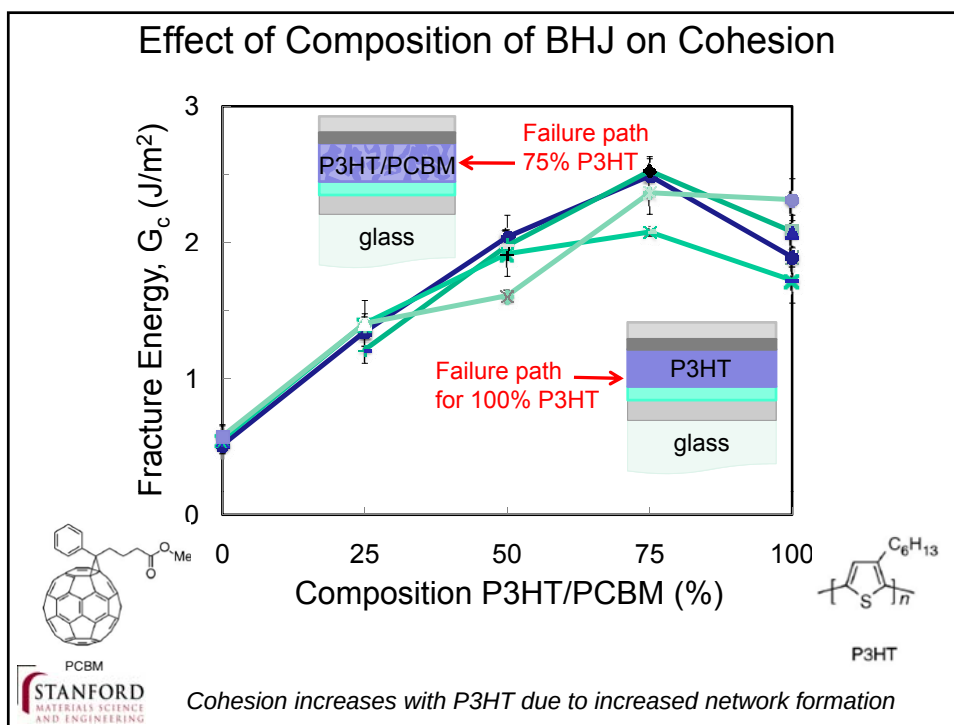
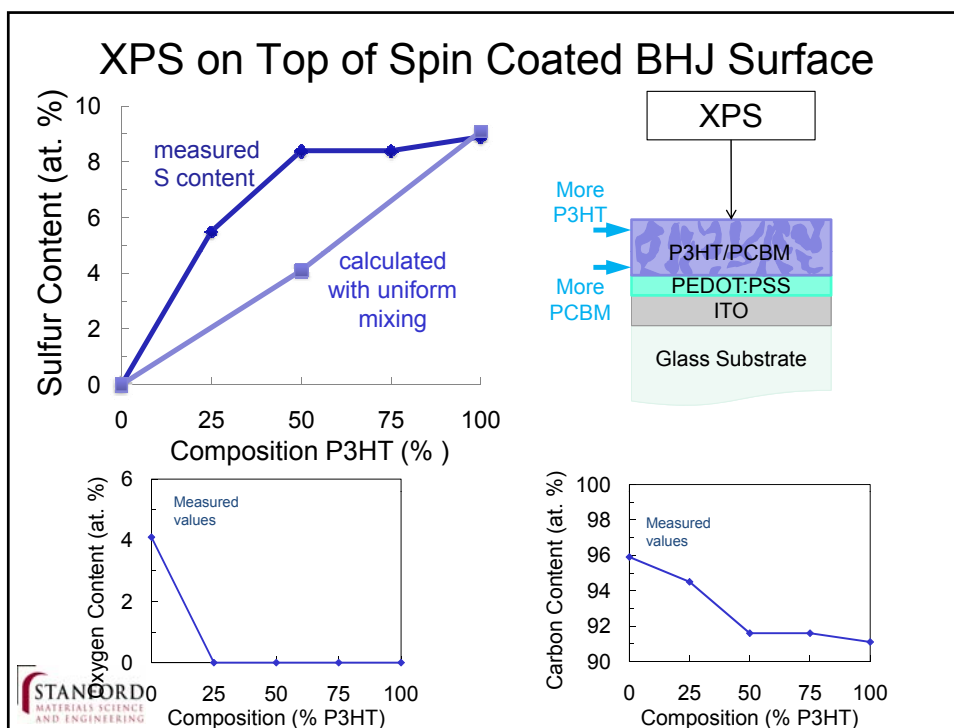
$$G_{\text{total}} = G_o + G_{\text{zone}}$$

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ductile adjacent layer

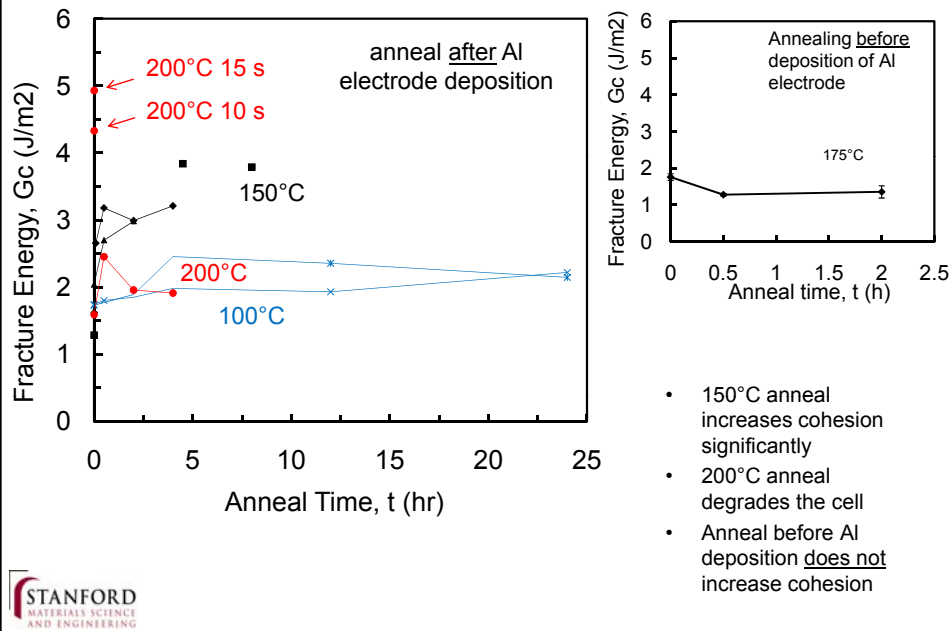


Segregation of PCBM towards bottom PEDOT due to more polar nature of PCBM compared to P3HT

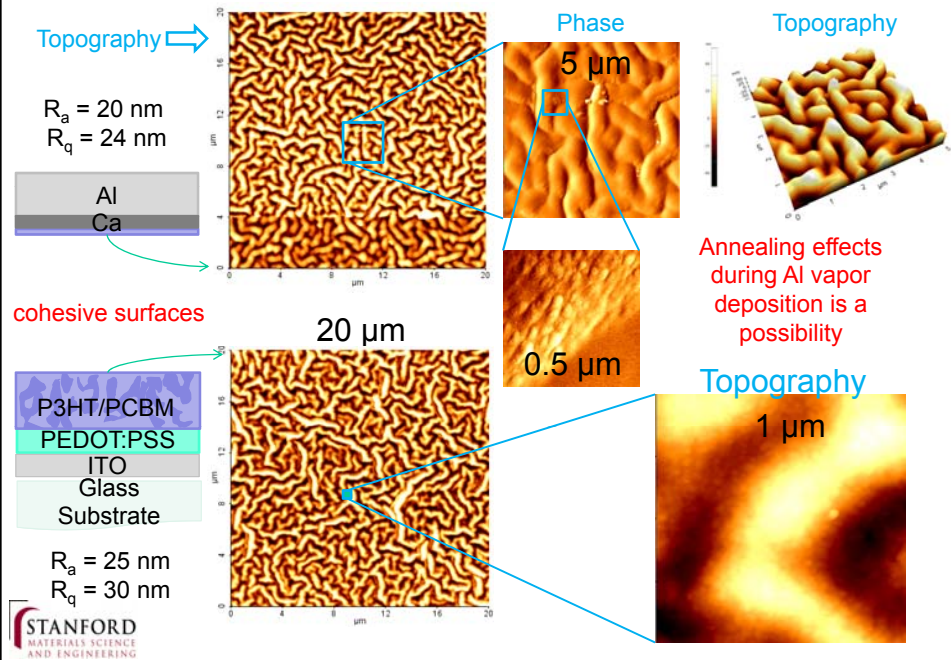




Effect of BHJ Annealing

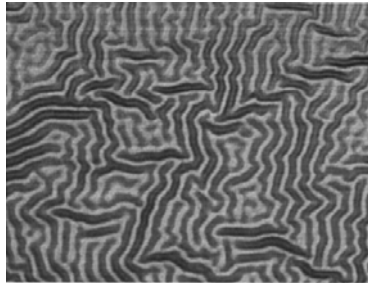


AFM of Failure Path Near Ca Interface



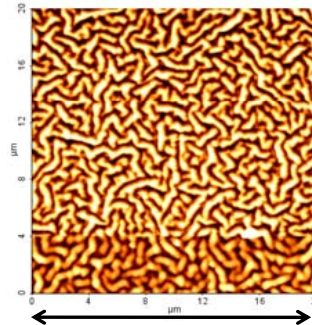
Buckling and Wrinkling in Stressed Films

Lacour et al. Appl. Phys. Lett., Vol. 82, No. 15, 14 April 2003



100 μm

Optical image of the wave pattern formed in a 100-nm-thick gold film evaporated on a 1-mm-thick PDMS membrane

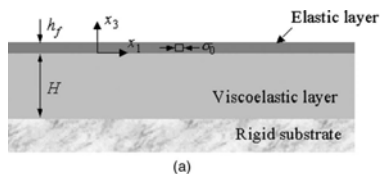


20 μm

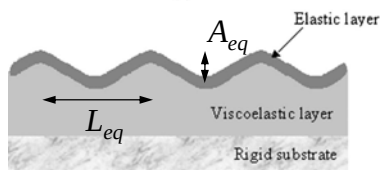
Buckling/wrinkling instability at the BHJ/Ca interface resulting from the vapor deposition of Al



Wrinkling of Stressed Elastic Film on Viscoelastic Layer



(a)



(b)

Equilibrium amplitude of sinusoidal wrinkle:

$$A_{eq} = \frac{2\sqrt{1-\nu_f^2}}{k} \left[-\frac{\sigma_0}{E_f} - \frac{(kh_f)^2}{12(1-\nu_f^2)} - \frac{2(1-\nu)}{1-2\nu} \frac{\mu_\infty}{E_f} \frac{1}{k^2 H h_f} \right]^{1/2}$$

Equilibrium wrinkle wavelength: $k = 2/L_{eq}$

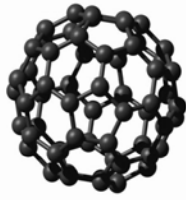
$$L_{eq} = \pi h_f \left[\frac{2(1-2\nu)}{3(1-\nu)(1-\nu_f^2)} \frac{E_f H}{\mu_\infty h_f} \right]^{1/4}$$

- Layers initially flat - elastic layer with in-plane biaxial compressive stress σ_0 .
- Wrinkling to relax stress in elastic layer – viscoelastic layer deforms to maintain conformality.
- μ_∞/E_f is ratio of rubbery modulus of the viscoelastic layer and the elastic modulus of the elastic layer.
- k is the wavevector: $k = 2/L_{eq}$

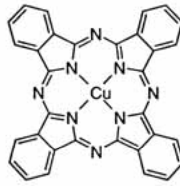


Huang et al. Journal of App. Mech. Nov. 2005, Vol. 72

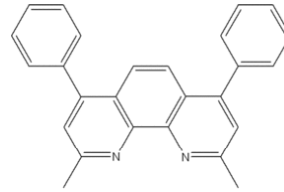
Small Molecule Solar Cell Thin Films



C60

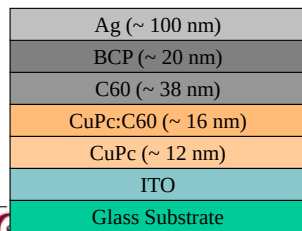


CuPc

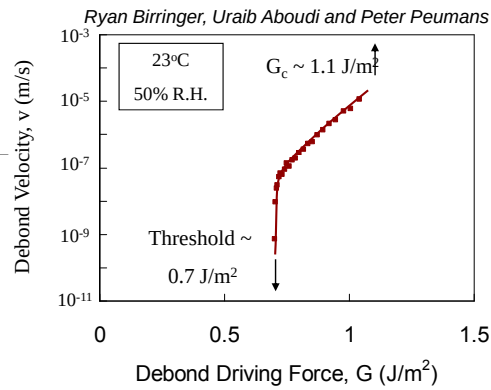


BCP

$$G_c = 1.1 \pm 0.4 \text{ J/m}^2$$



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Molecular Bond Rupture Kinetics (Barrier Films)

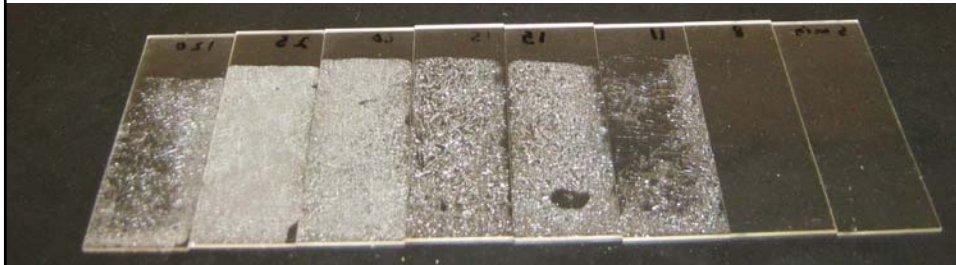
Grad student: Fernando Novoa and Monika Kummel

$$G < G_c \left[\text{J/m}^2 \right] \quad \text{environment and stress accelerates defect evolution}$$

Weathering Test of Polysiloxane Barrier

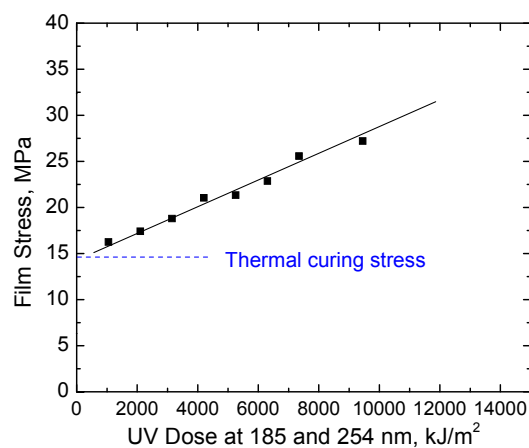
UV exposure: 28 mW/cm² at 6 mm UV-257nm

120 min.....15 min.....5 min



Weathering Test of Polysiloxane Barrier

Film Stress Accumulation with UV Exposure



Driving force for damage:

$$G = \frac{Z \sigma_f^2 h_f}{E_f}$$



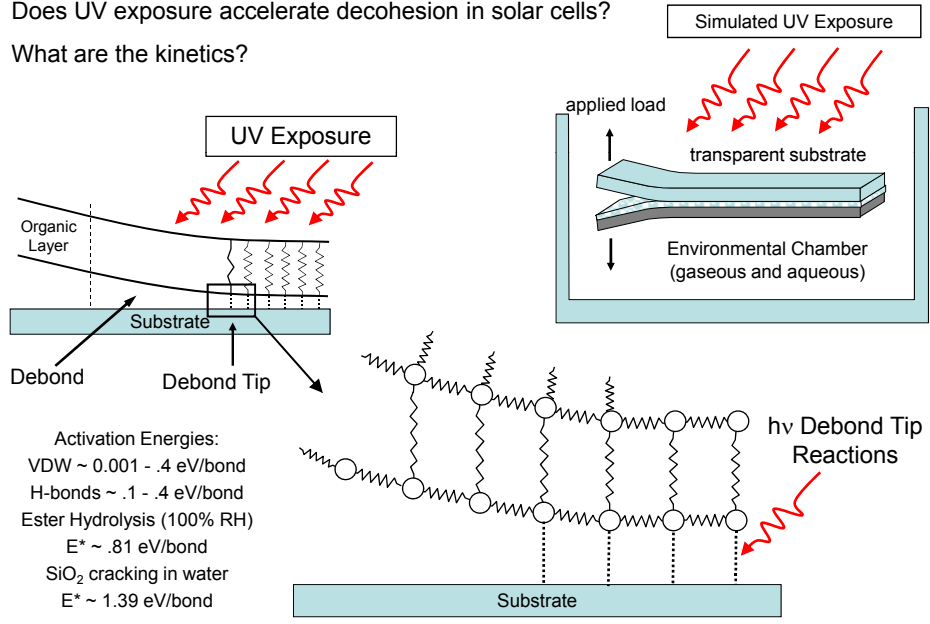
Kamer and Dauskardt - 2009



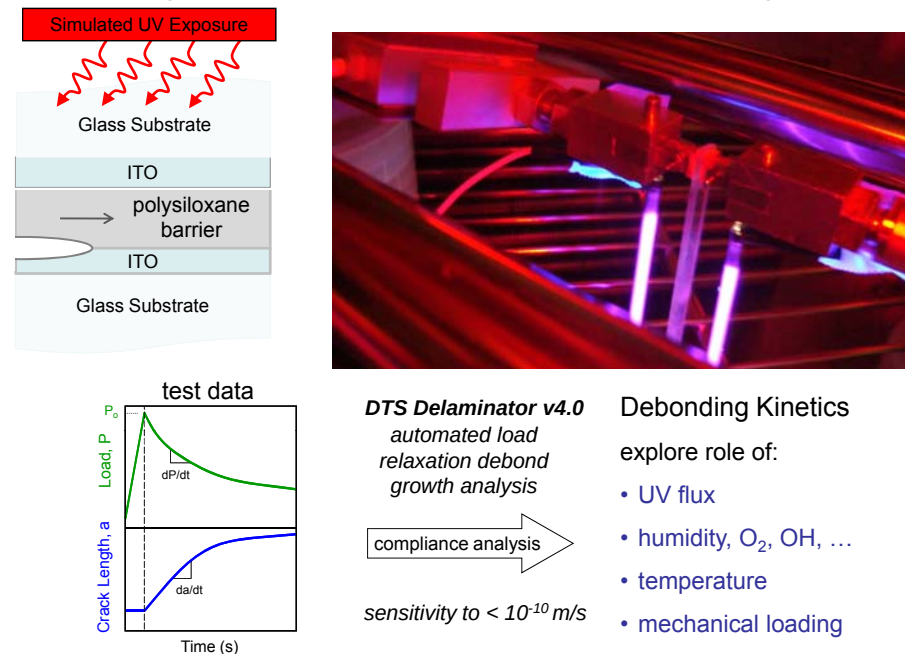
Environment and Stress Accelerates Damage

Does UV exposure accelerate decohesion in solar cells?

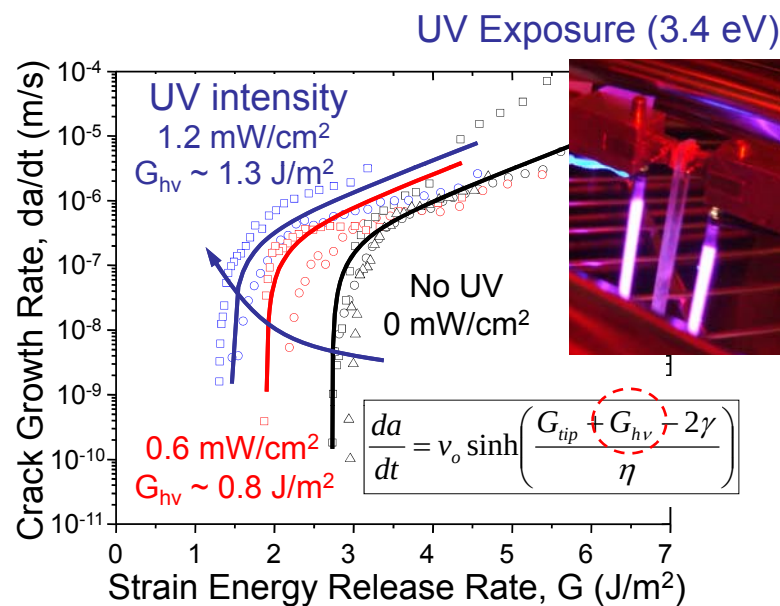
What are the kinetics?



Assessing UV and Environment on Debonding Kinetics

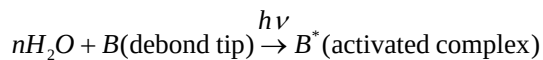


UV Effects on Molecular Bond Rupture



Modeling Bond Rupture Kinetics

- Interaction of moisture with strained debond tip bonds



- Atomistic bond rupture models:

$$\text{rate} = f_o \left[\exp\left(\frac{-U_+}{kT}\right) - \exp\left(\frac{-U_-}{kT}\right) \right]$$

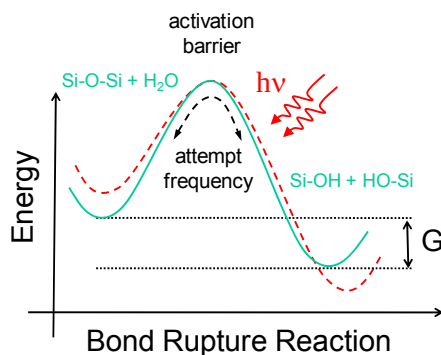
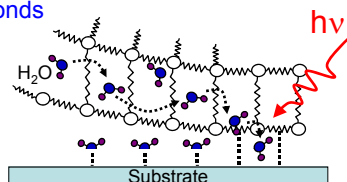
- Damage growth rate:

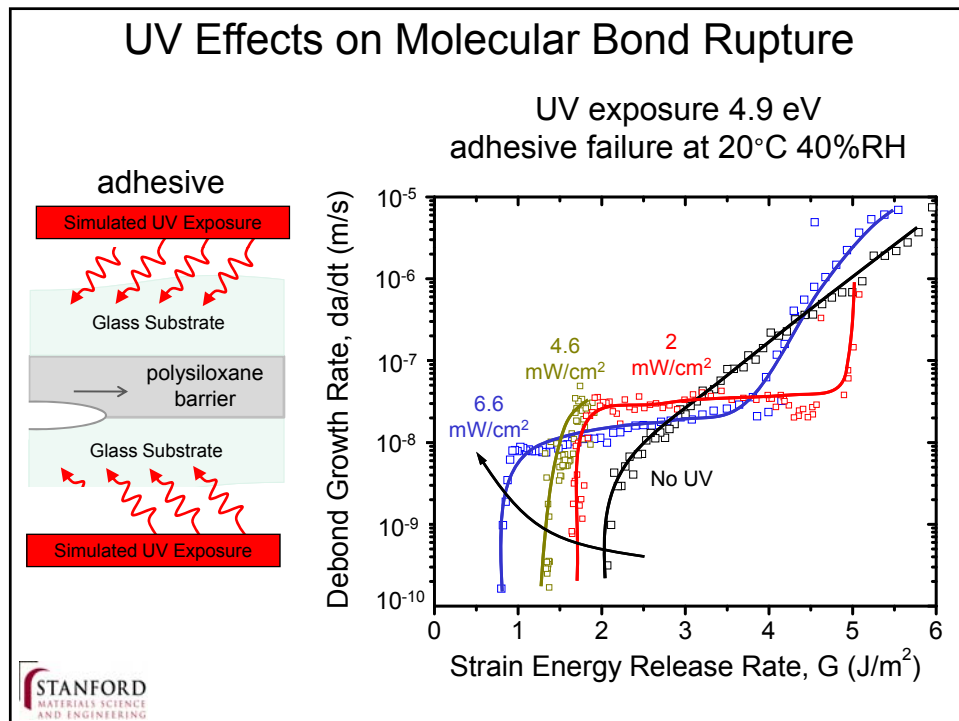
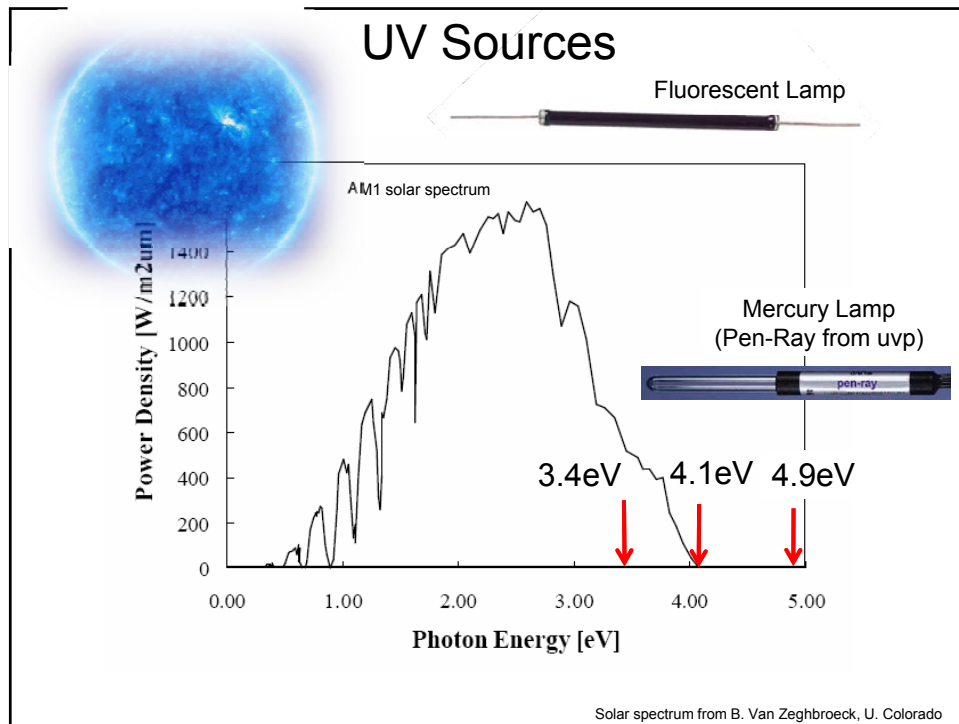
$$\frac{da}{dt} = v_o \sinh\left(\frac{G_{tip} + G_{hv} - 2\gamma}{\eta}\right)$$

$$v_o = \frac{2 f_o}{Nw} \exp\left(\frac{-u_1}{kT}\right)$$

Bond Rupture Parameters

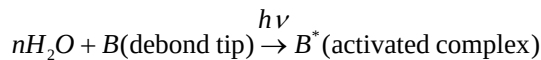
N - bonds per unit area f_o - attempt frequency
 u_o - work of rupture u_1 - energy barrier
 2γ - $N u_o$ η - $2NkT$
 w - crack width





Modeling Bond Rupture Kinetics

- Interaction of moisture with strained debond tip bonds



- Atomistic bond rupture models:

$$\text{rate} = f_0 \left[\exp\left(\frac{-U_+}{kT}\right) - \exp\left(\frac{-U_-}{kT}\right) \right]$$

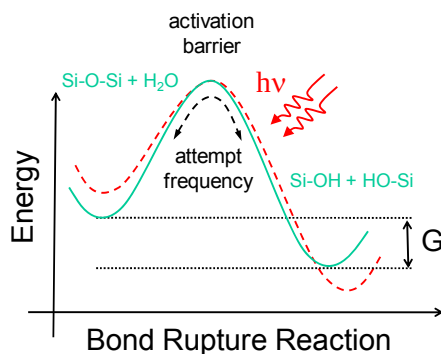
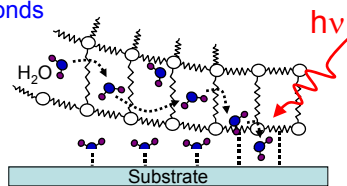
- Damage growth rate:

$$\frac{da}{dt} = v_0 \sinh\left(\frac{G_{tip} + G_{hv} + 2\gamma}{\eta}\right)$$

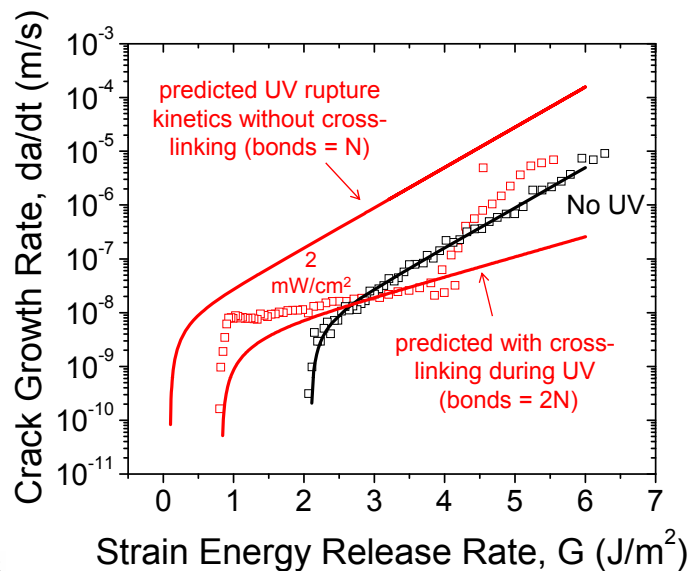
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Bond Rupture Parameters

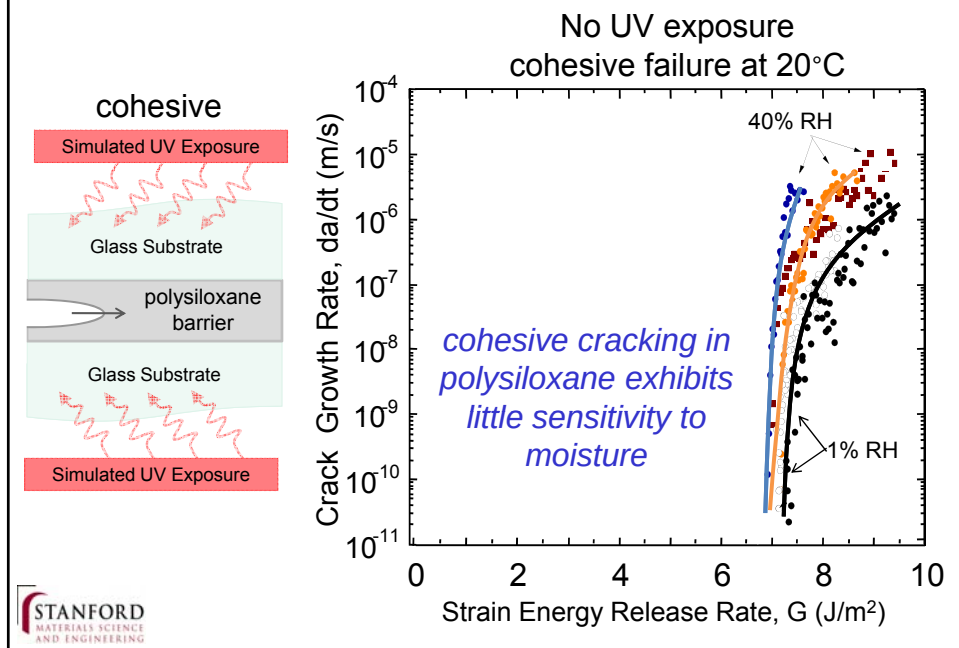
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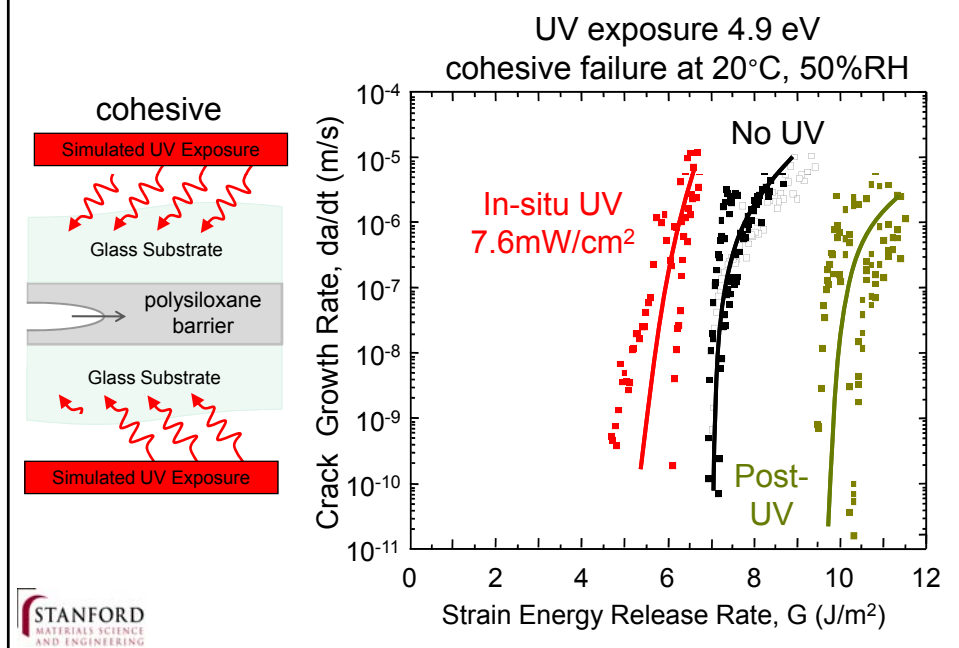
Modeling Bond Rupture Kinetics



UV Effects on Molecular Bond Rupture



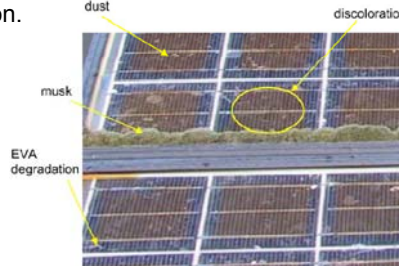
UV Effects on Molecular Bond Rupture



Delamination of EVA-TPE Lamination

- Poly-ethylene vinyl acetate (EVA) copolymer extensively used by solar module manufacturers, particularly for laminating c-Si photovoltaic modules.
- Good optical properties and high adhesive contact with glass cover and Si cells.
- Inexpensive and relatively easy fabrication.

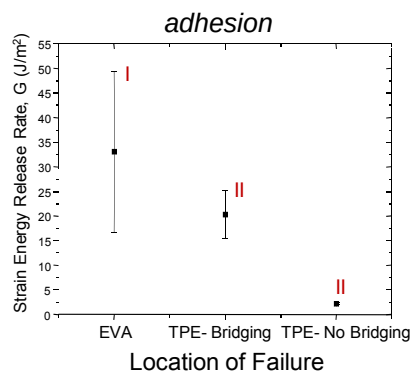
Parreta Antonio, et al., Solar Energy Materials & Solar Cells, 2005



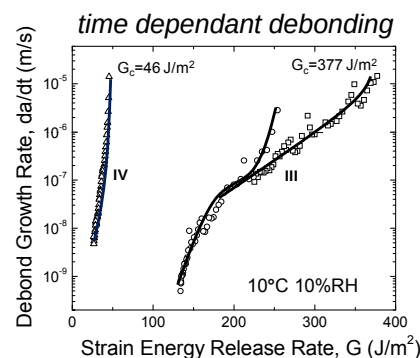
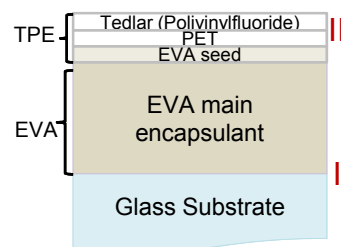
- Delamination can occur between EVA and the front surface of the solar cells.
- More frequent and in hot and humid climates.
- Exposure to atmospheric water and/or ultraviolet radiation leads to EVA decomposition to produce acetic acid, lowering the pH and increasing corrosion.
- EVA $T_g \sim -15^\circ\text{C}$ so lower temperatures may result in “ductile-to-brittle” transition in adhesive/cohesive properties.



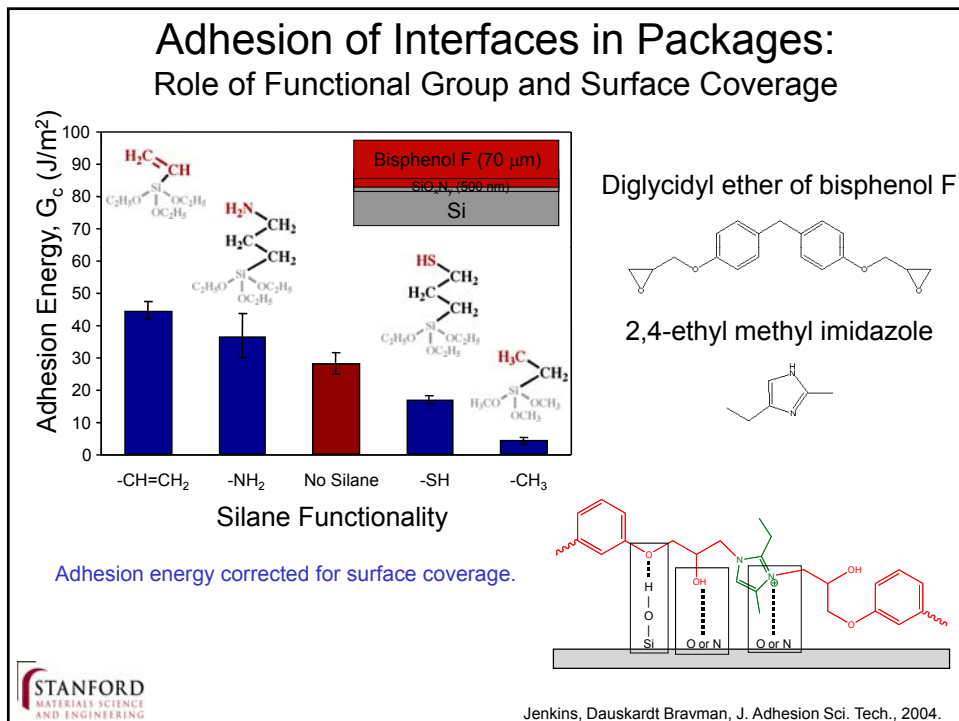
Delamination of EVA-TPE Lamination



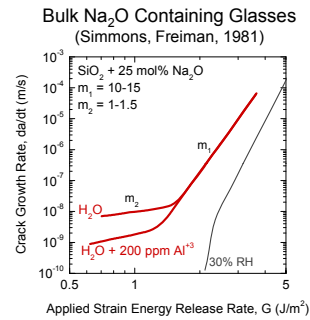
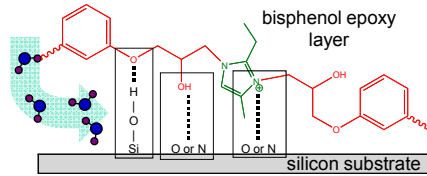
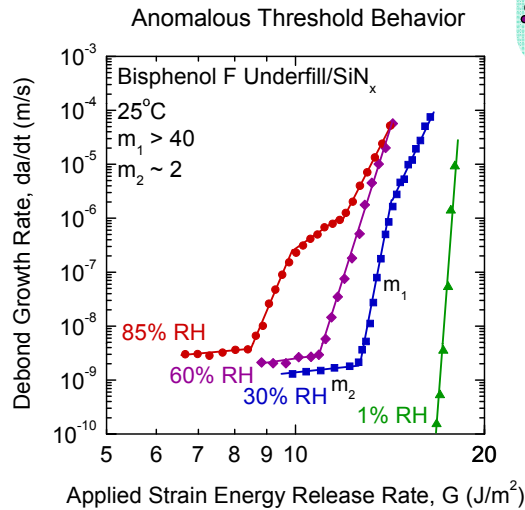
Interface “I” located between EVA and Glass
Interface “II” inside the TPE multilayer



Examples from Microelectronics



Adhesion and Debonding in Device Packaging



Wang, Sharratt and Dauskardt, Acta Materialia – 2007.

Stress-Dependent Moisture Transport Model

$$D = D_0 \exp(k_2 C + k_3 \sigma)$$

Petropoulos & Roussis (1978)

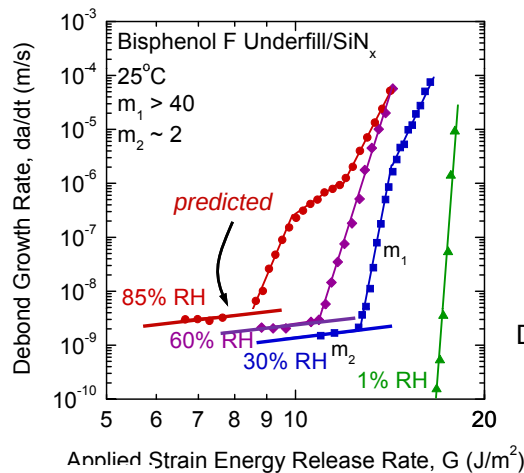


$$D = D_0 \exp\left(\frac{-E^* + \alpha\sqrt{G}}{RT}\right)$$



$$\frac{da}{dt} = \frac{c u_{H_2O}}{\delta n \Sigma} D_0 \exp\left(\frac{-E^* + \alpha\sqrt{G}}{RT}\right)$$

$$E^* \sim 25 \text{ kJ/mol}$$

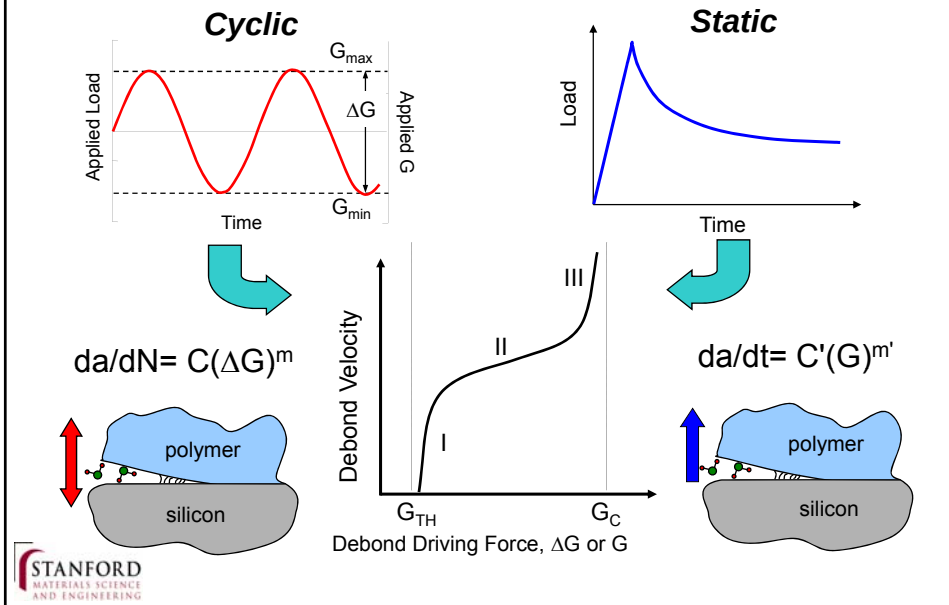


Diffusion in similar epoxy systems:

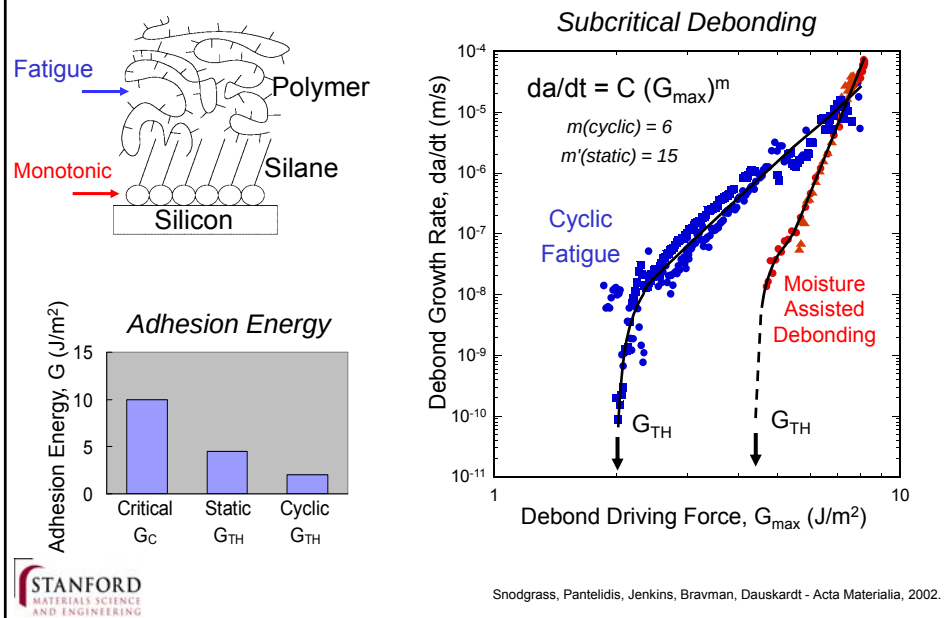
DGEBA/1,3-BAC (Barral et al, 1996)	31.2 kJ/mol
TGDDM/DDS (Musto et al, 2002)	34 kJ/mol

Wang, Sharratt and Dauskardt, Acta Materialia – 2007.

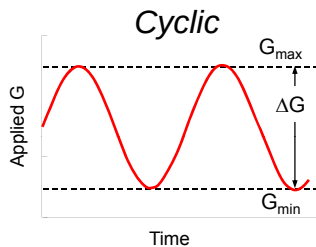
Time Dependent Debonding: Cyclic and Static Loading



Polymer(BCB)/Silica Subcritical Debonding



Cyclic Stress-Dependent Transport Model

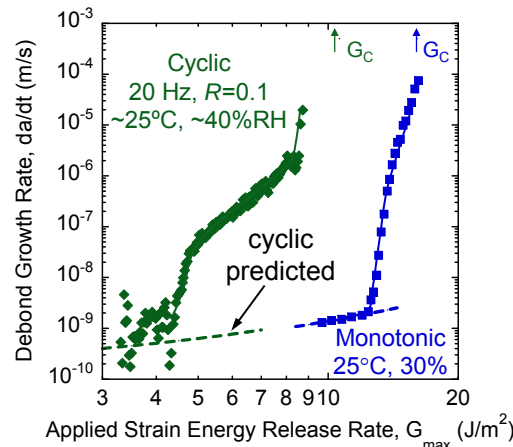


monotonic

$$\frac{da}{dt} = \frac{cua_{H_2O}}{\delta n \Sigma} D_0 \exp\left(\frac{-E_d + \alpha\sqrt{G}}{RT}\right)$$

cyclic (integrate stress-dependent diffusion over fatigue loading cycle)

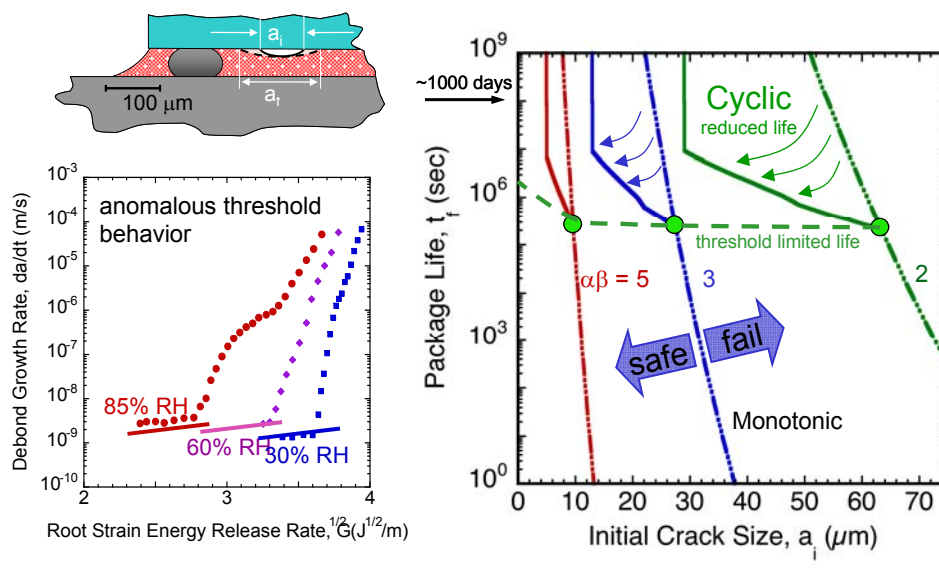
$$\frac{da}{dt} = \frac{cua_{H_2O}}{\delta n \Sigma} D_0 \exp\left(\frac{-E_d}{RT}\right) g(f, t)$$



Wang, Sharratt and Dauskardt, Acta Materialia – 2007.

Effect of Fatigue and Thresholds on Package Life

- Significant reduction in life under cyclic loading indicated by arrows.
- Further limits on package life result from anomalous thresholds



Our Goals for Reliability of PV Technologies

We want to engineer reliable PV devices and modules with robust and predictable lifetimes.

- Leverage from reliability physics in microelectronics – mechanisms, kinetic models, accelerated tests and life prediction
- Develop metrologies to quantitatively characterize thermo-mechanical properties (e.g. adhesion, cohesion), photochemical and environmental degradation processes
- Are degradation processes coupled and how?
- Kinetic models of damage evolution - basis for life prediction and accelerated testing (effect of operating temperature, environment, mechanical stress, solar flux, etc.)
- Effective transparent barriers with anti-reflective properties and low cost.