

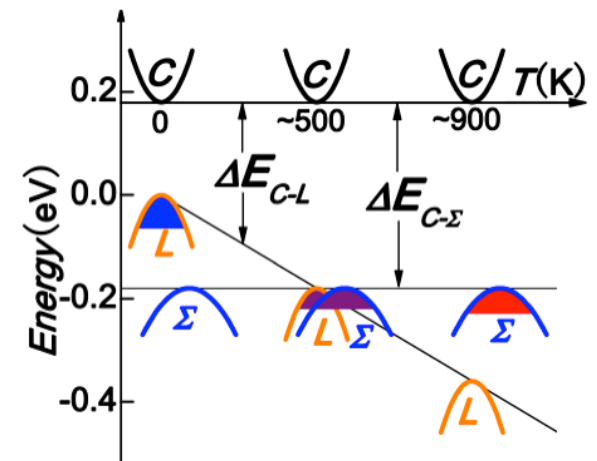
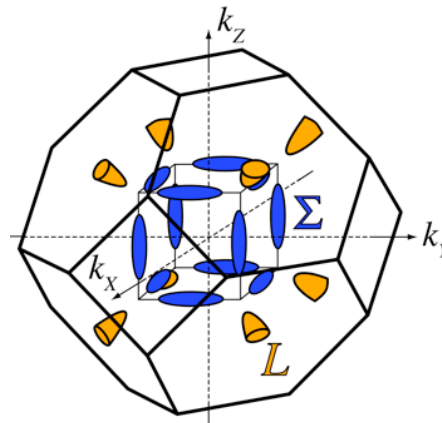
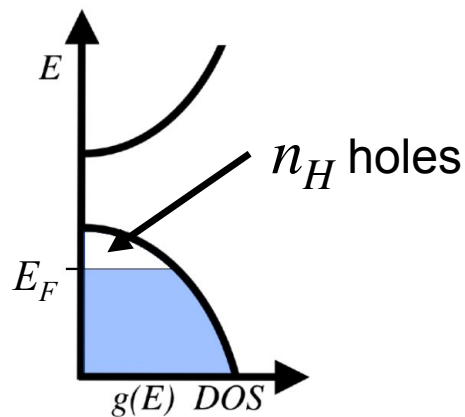
Materials strategies for improving the overall device ZT

G. Jeffrey Snyder

California Institute of Technology

Pasadena, California, USA

<http://thermoelectrics.caltech.edu>





Carrier Concentration

Desire High zT Figure of Merit

$$zT = \frac{\alpha^2 \sigma T}{\kappa}$$

Conflicting Materials Requirements

α Seebeck Coefficient

Need small n , large m^*

- Semiconductor (Valence compound)

$$\alpha = \frac{8\pi^2 k_B^2}{3eh^2} m^* T \left(\frac{\pi}{3n} \right)^{2/3}$$

σ Electrical Conductivity

Need large n , high μ

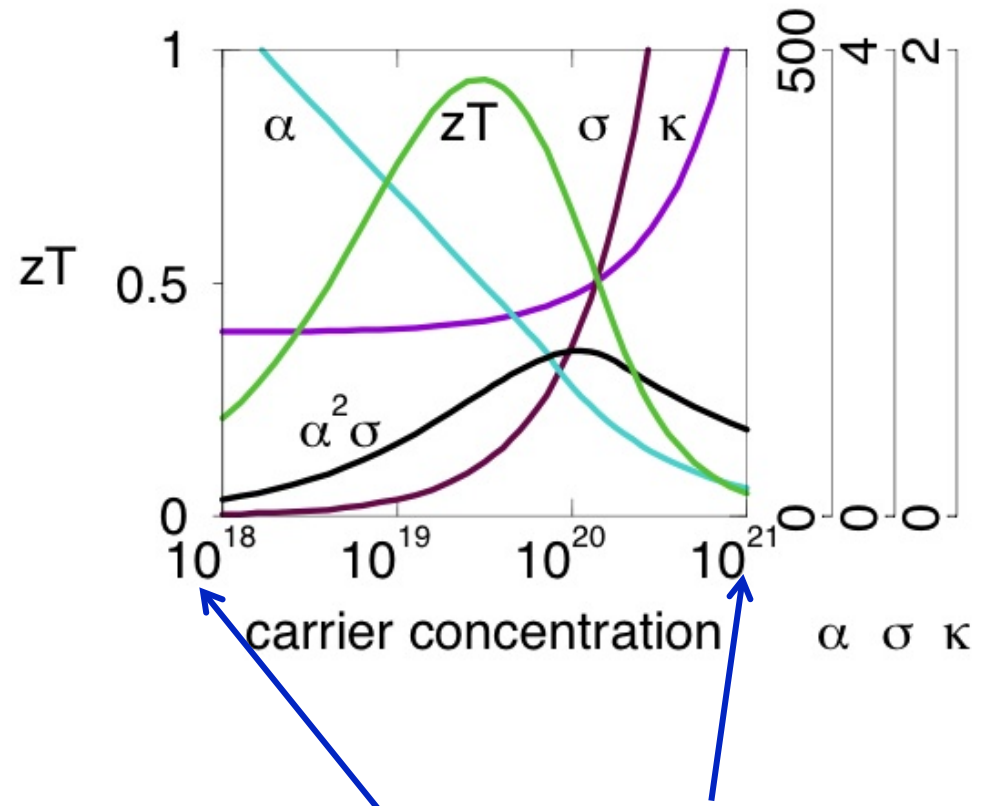
- Metal

$$\sigma = ne\mu$$

κ Thermal Conductivity

Desire small κ_l , small n

$$\kappa \approx \kappa_l + LTne\mu$$



Optimum between Insulator and Metal



Thermoelectrics
Caltech Materials Science

Snyder, Toberer *Nature Materials* 7, 105 (2008)

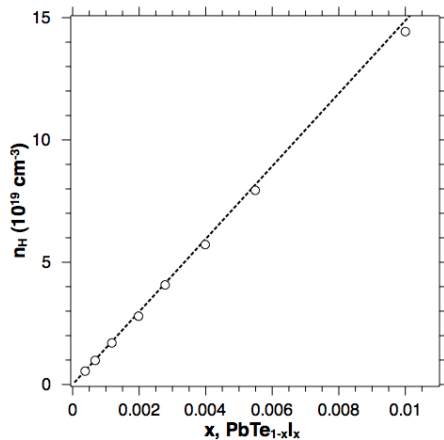
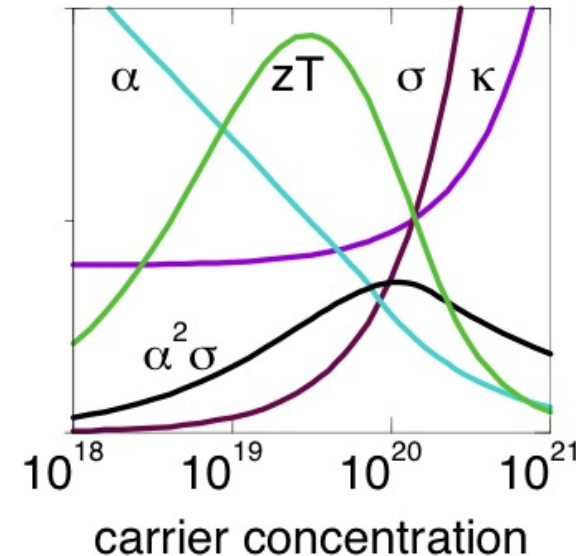
Carrier Concentration Tuning



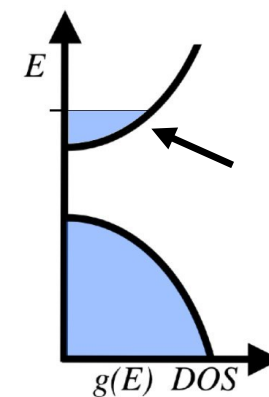
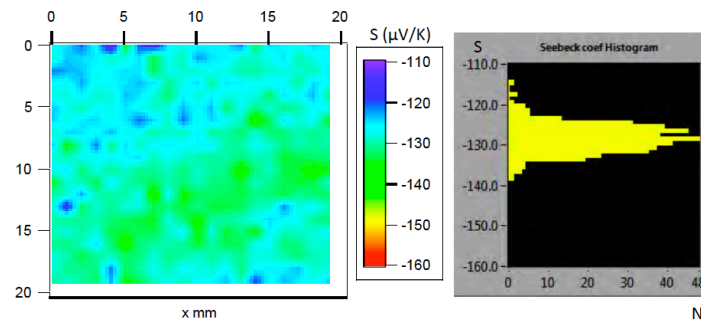
Iodine (I) supplies one more electron than Tellurium (TE)

50 Sn Tin 118.710	51 Sb Antimony 121.760	52 Te Tellurium 127.60	53 I Iodine 126.904 47
-----------------------------------	--	--	--

Iodine (I-) replaces Te^{2-}
producing $1 e^-$ $10^{18} - 10^{20} e^-/\text{cm}^3$



From Room Temperature Hall Effect



I
ons
und

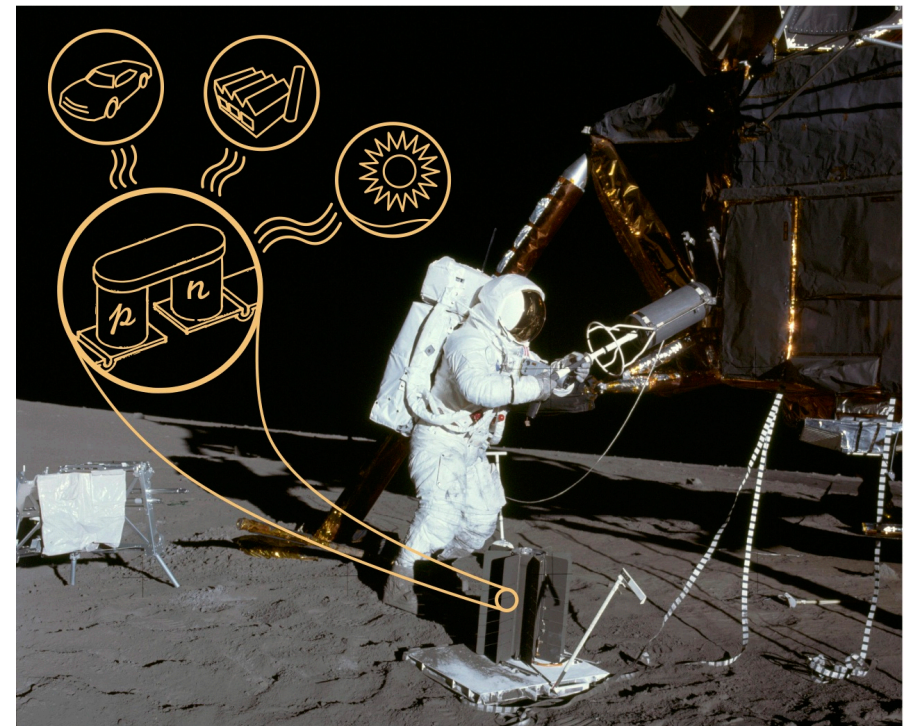
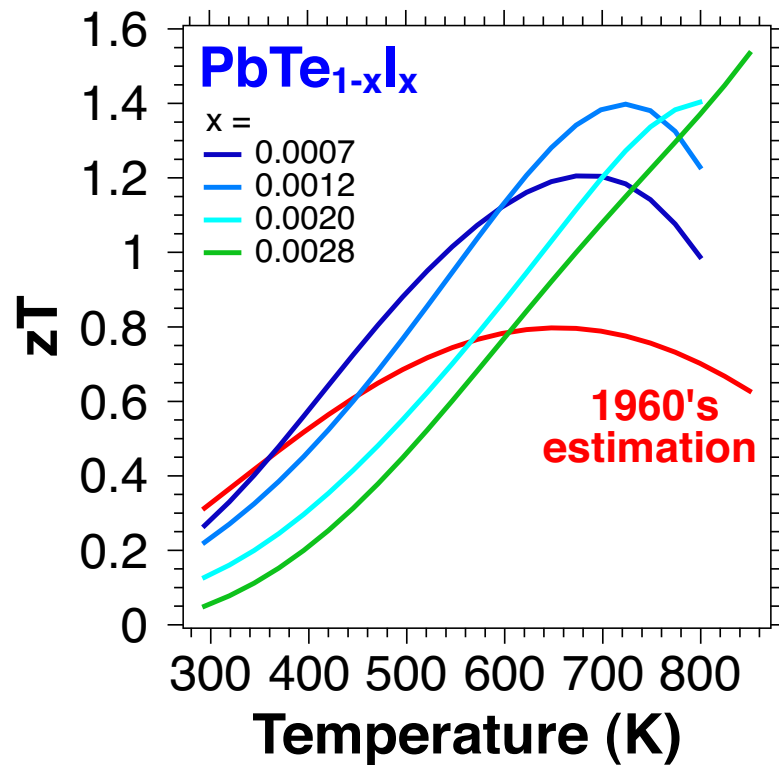


Thermoelectrics
Caltech Materials Science

zT of n-type $\text{PbTe}_{1-x}\text{I}_x$



Peak $zT \sim 1.4$ at 800K due to low thermal conductivity



Unattended operation for over 10 years



Thermoelectrics
Caltech Materials Science

Lalonde, Pei, Snyder *Energy and Environmental Science* 4, 2090 (2011)

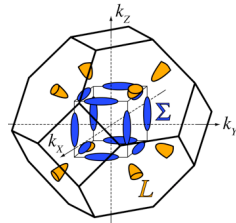
Quality Factor *and* n^*

Maximum zT depends on
Quality Factor

$$B \sim \frac{N_V}{m_b^* K_L}$$

Multi Valley Fermi Surface
with Valley Degeneracy N_V

$$m^* = m_b^* N_V^{2/3}$$



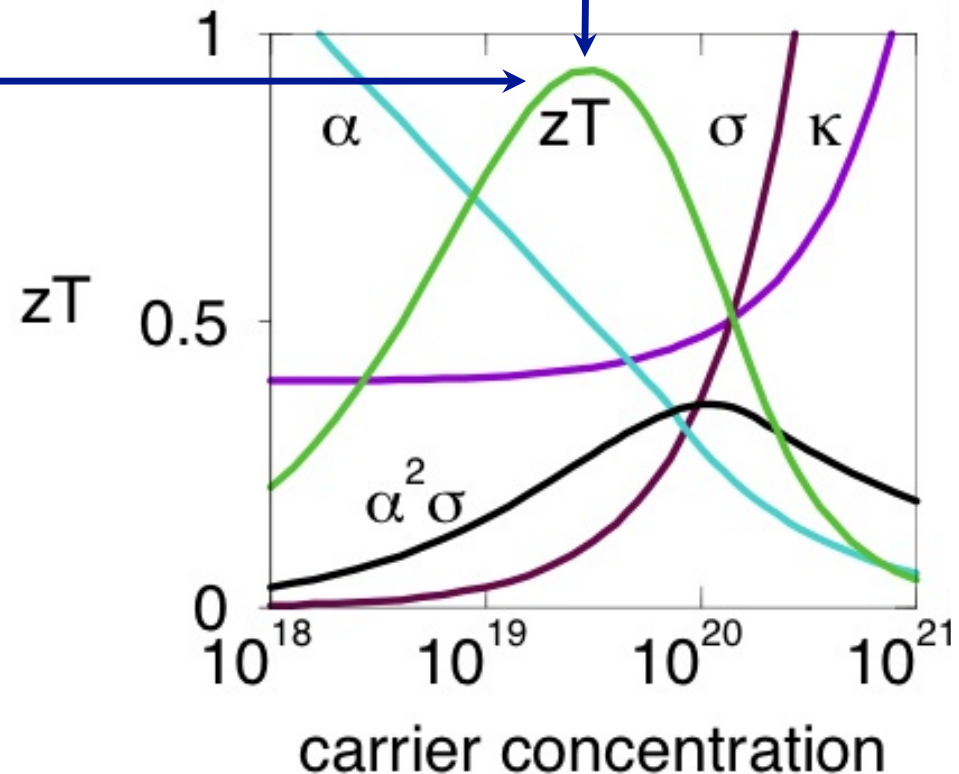
because μ decreases with m^*

$$\mu = \frac{e\tau}{m_b^*} \quad \tau \propto \frac{1}{m_b^{*3/2}}$$

Acoustic Phonon Scattering

Optimized
carrier concentration

$$n^* \sim N_V^{2/3} m_b^* T^{3/2}$$

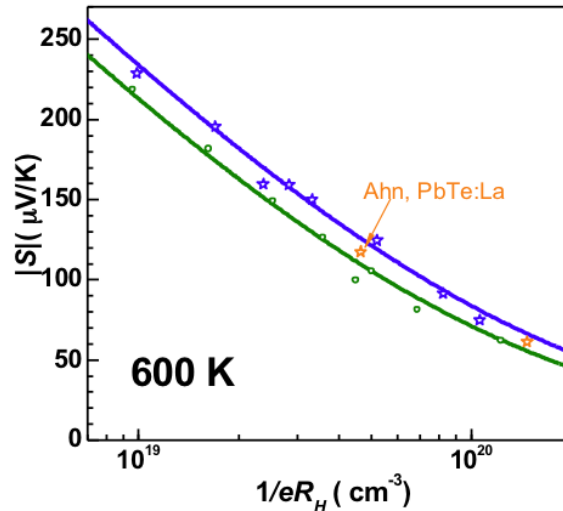


Effective mass

$\text{La}_x\text{Pb}_{1-x}\text{Te}$ vs. $\text{PbTe}_{1-x}\text{I}_x$

Both n-type L-band

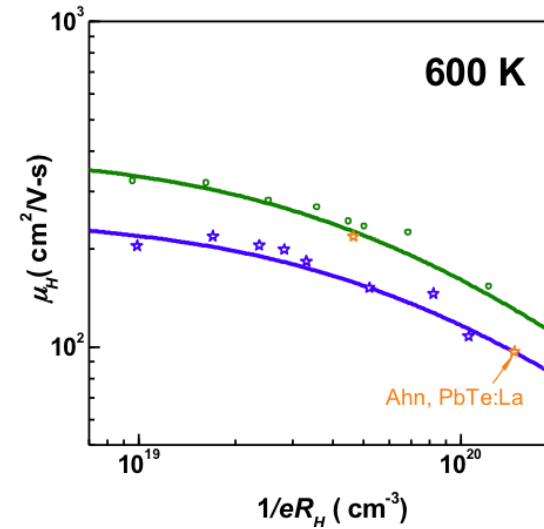
20% lower m^*



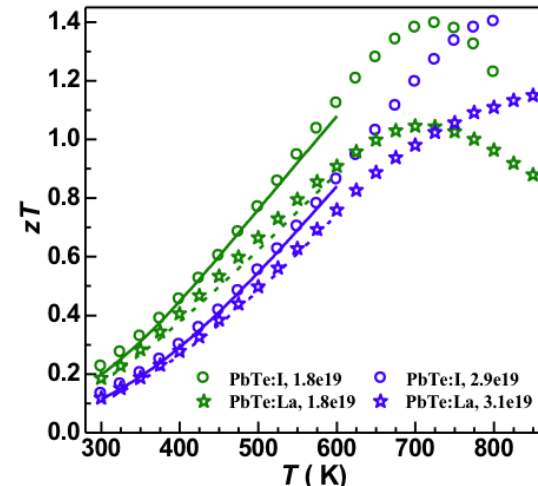
$$\mu \propto \frac{1}{m_b^{*5/2}}$$

$$B \sim \frac{N_V}{m_b^* K_L}$$

30% higher μ



20% Higher zT



Heavy and Light holes in PbTe

Valence Band Maximum is at L point

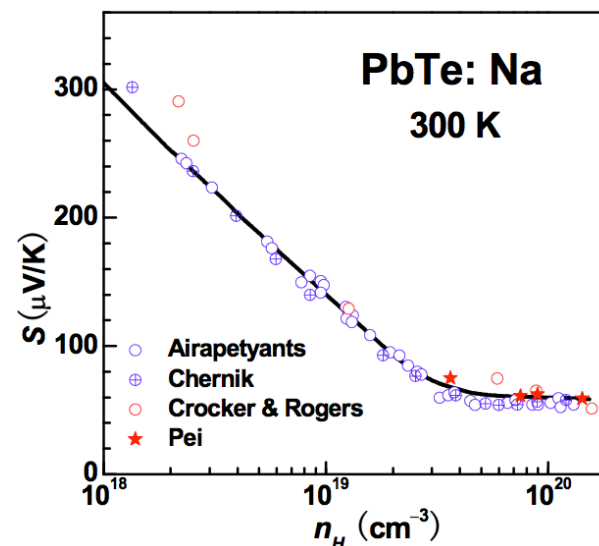
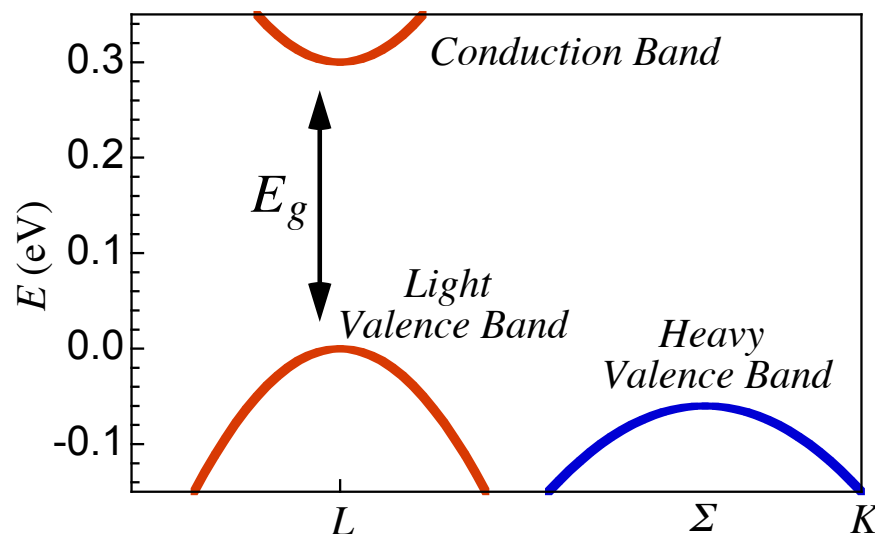
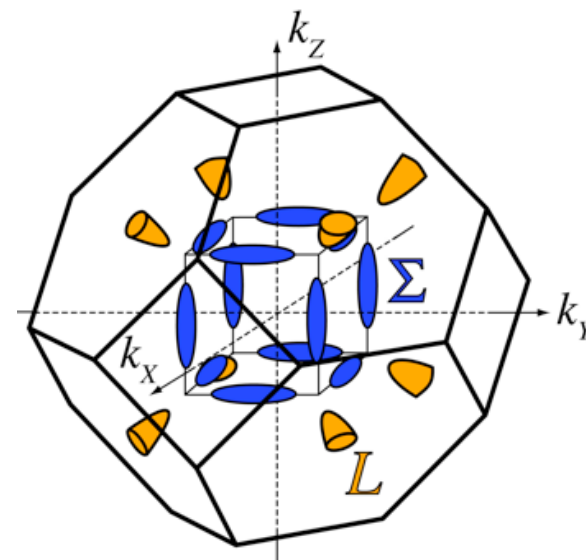
- “Light Band” $N_V = 4$, $m_b^* = 0.14 m_e$

Second valence band occurs at Σ line

- “Heavy Band” $N_V = 12$, $m_b^* = 0.28 m_e$

$$m^* = m_{band}^* N_V^{2/3}$$

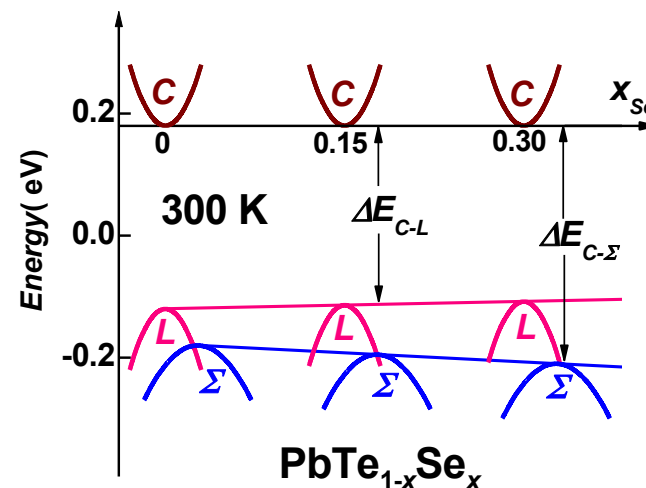
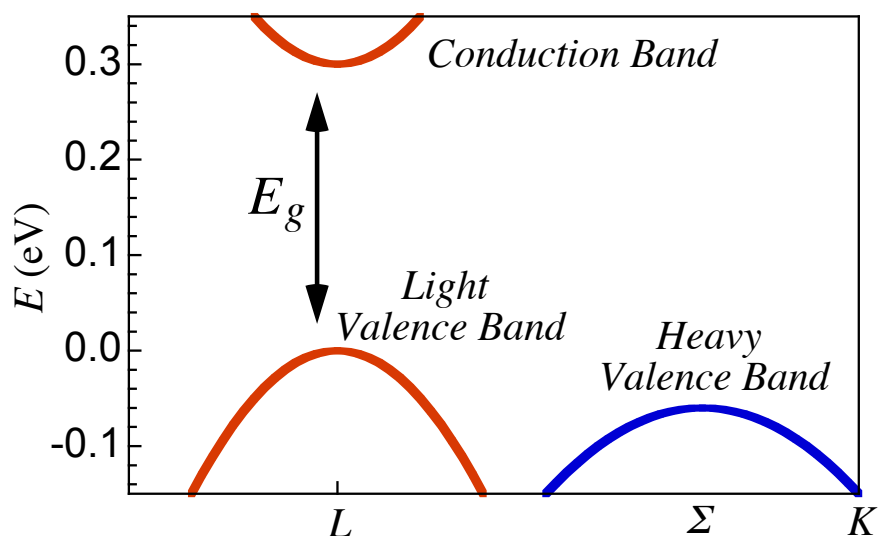
Transition from single to multiple band occurs at $n_H \sim 3 \times 10^{19}$ holes/cm³



Valley Convergence with composition

PbTe convergence changes with alloying

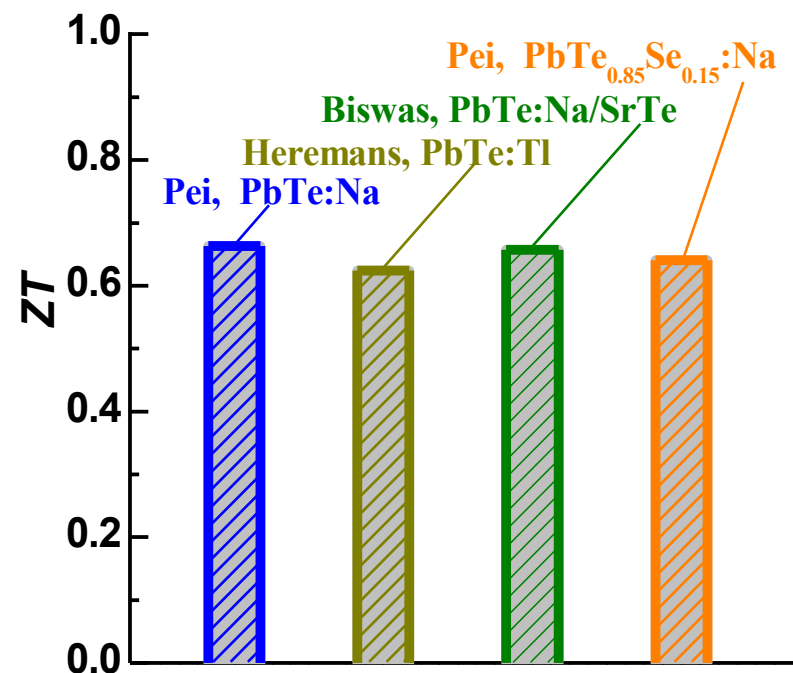
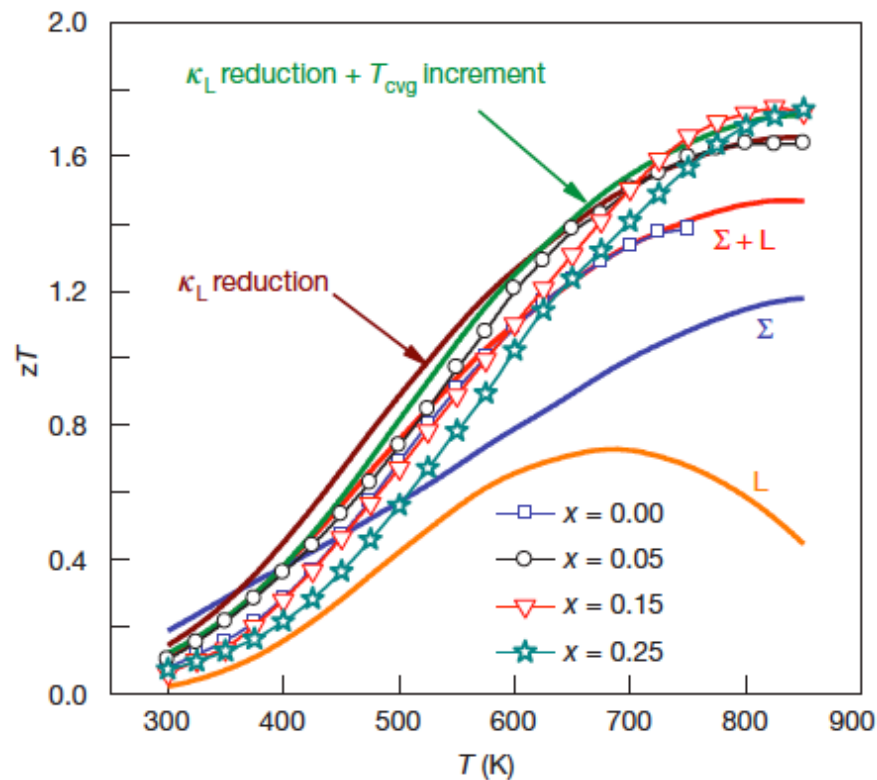
- Slight shift in energy of C, L and Σ bands
- Will change Convergence Temperature
 - Increases T_{convg} with PbSe alloying
 - Decreases T_{convg} with MgTe alloying



p-type PbTe-PbSe ZT (average zT)



Peak zT is high but device ZT (average zT) is not really improved
 zT at low temperature reduced by increasing T_{conv} !



zT vs. ZT

Materials figure of merit zT

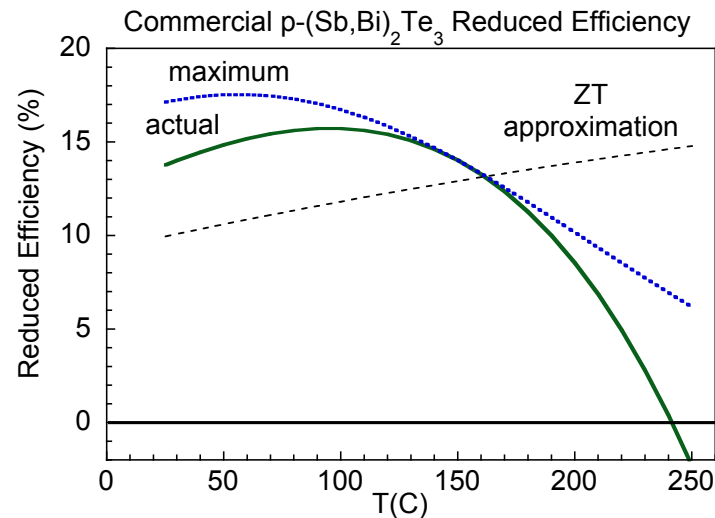
$$zT = \frac{\alpha^2}{\rho\kappa} T$$

Determines maximum reduced efficiency at any given point

$$\max \eta_r = \frac{\sqrt{1 + zT} - 1}{\sqrt{1 + zT} + 1}$$

Actual efficiency is less

- series current not optimal everywhere



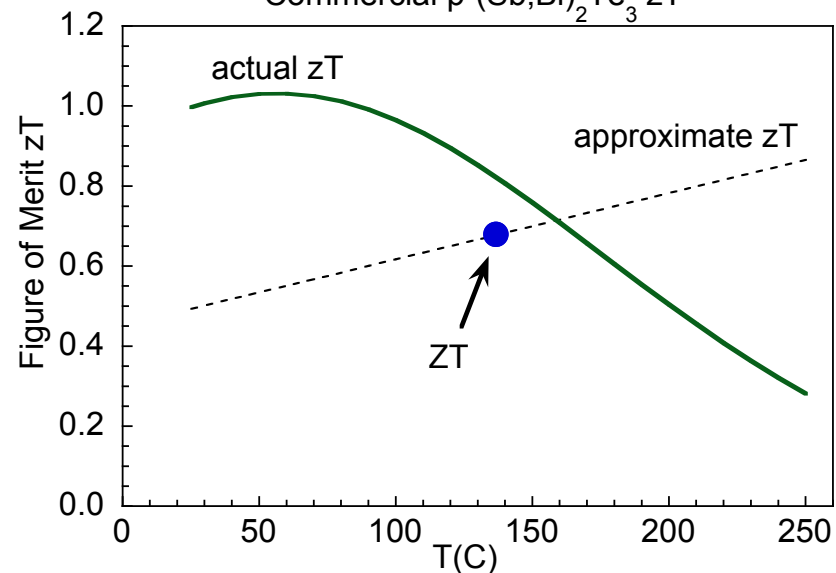
Device Figure of Merit ZT

$$\eta = \frac{\Delta T}{T_h} \cdot \frac{\sqrt{1 + ZT} - 1}{\sqrt{1 + ZT} + T_c/T_h}$$

$ZT = zT$ is approximation for

$\alpha(T)$, $\rho(T)$, $\kappa(T)$, are constant

Commercial p-(Sb,Bi)₂Te₃ zT



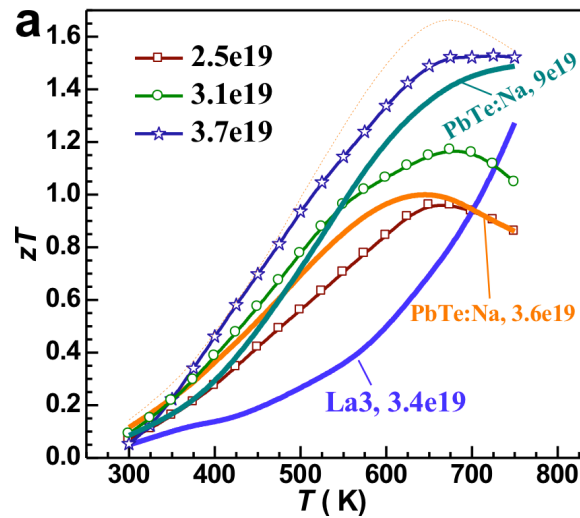
Real materials: $zT \neq ZT$

especially $\max zT \neq ZT$

Beware conclusions about changing ΔT

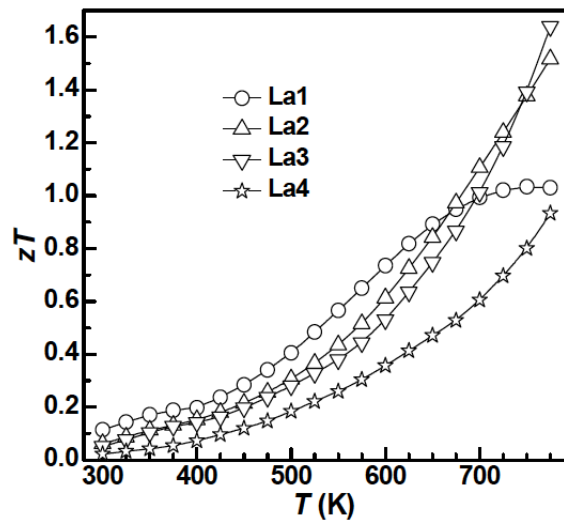
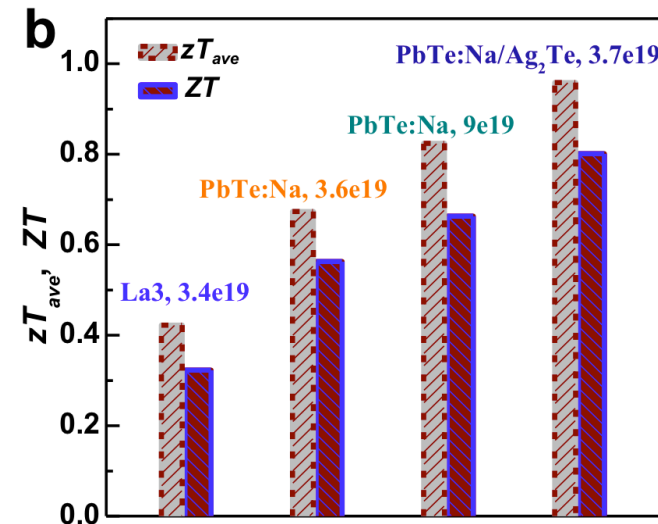


Precipitates in PbTe can Increase ZT

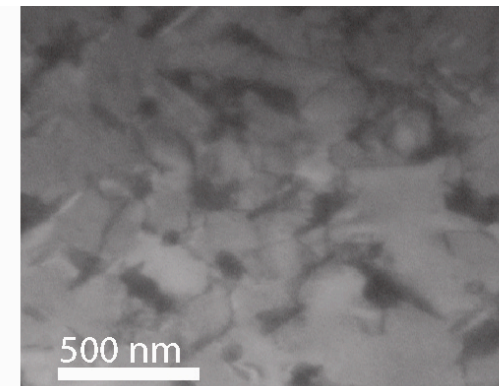


p-type

Higher average zT
Higher Device ZT



n-type



Thermoelectrics
Caltech Materials Science

Pei, Snyder, et al, *Energy and Environmental Science* 4, 3640 (2011)

Pei, Snyder, et al, *Adv. Funct. Mat.*, **21**, 241 (2011)

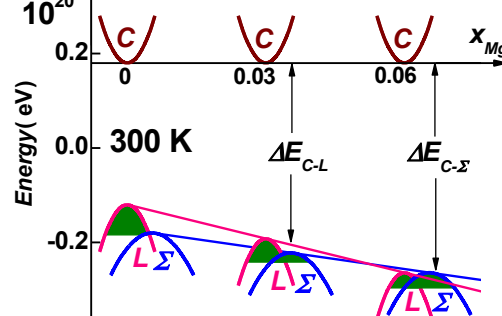
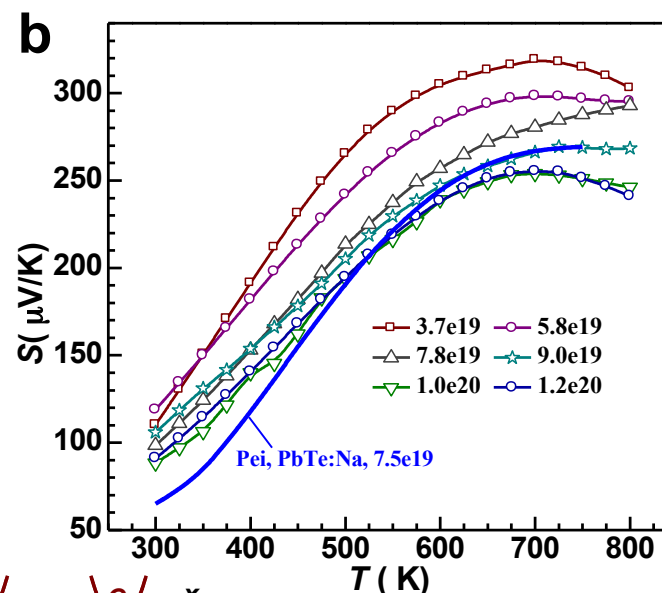
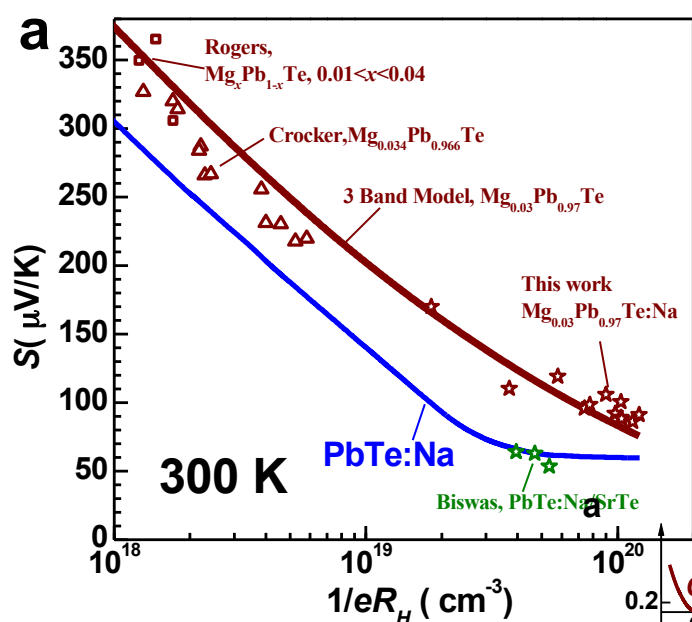
Decreasing T_{conv} with Mg

$\text{Mg}_x\text{Pb}_{1-x}\text{Te}$ shifts valence L band to convergence with Σ at 300K

S vs n (Pisarenko) looks like single heavy band

Continues to go below Σ at high temperature

But band gap (between C and Σ) increases, helps S at high T



Increasing ZT by stabilizing n^*

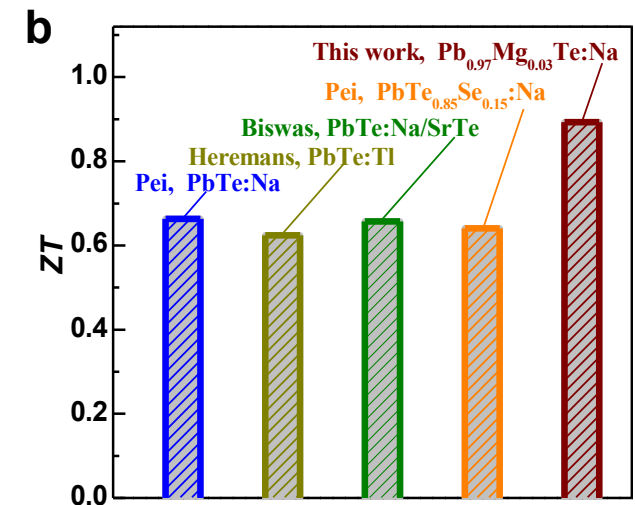
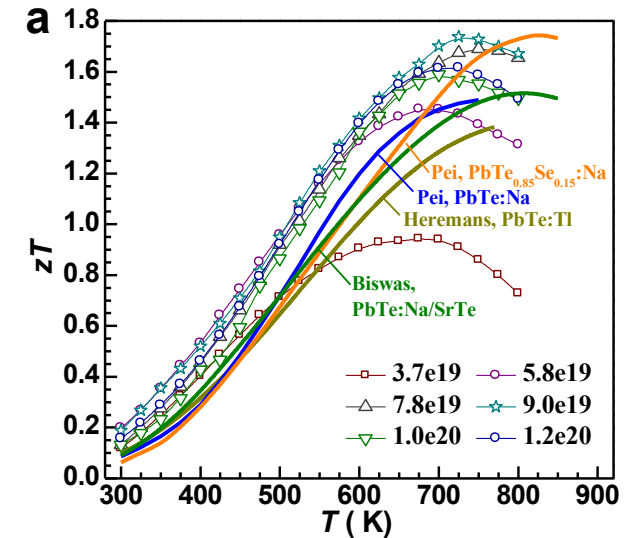
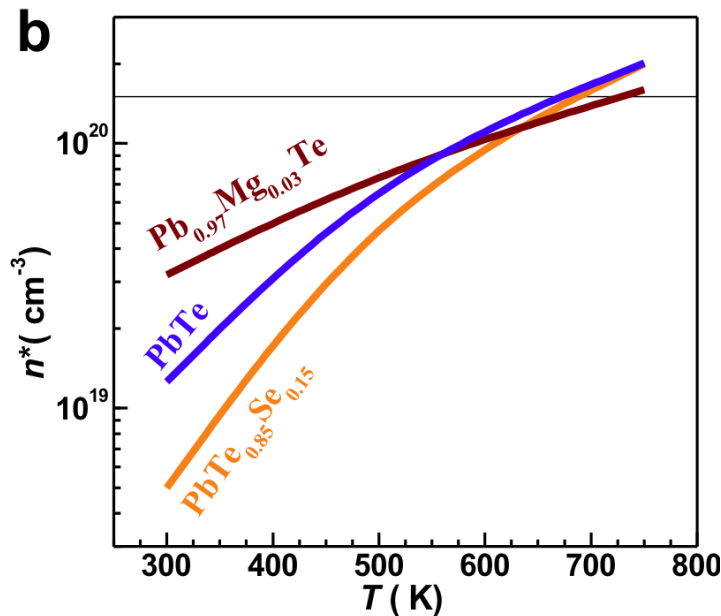
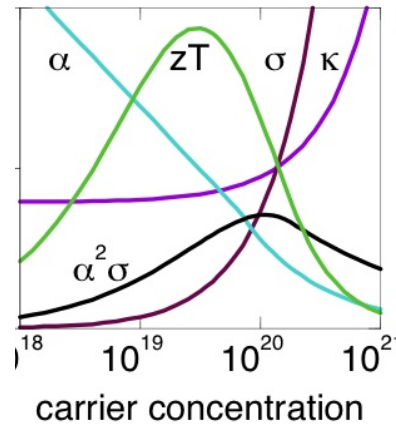
Lower T_{conv} :

Reduces T of N_V
enhancement

Keeps n^* from increasing
with T so fast

Higher average zT

Optimized
carrier
concentration
 $n \sim N_v^{2/3} m_b^* T^{3/2}$



Self-Tuning n^* via Composites

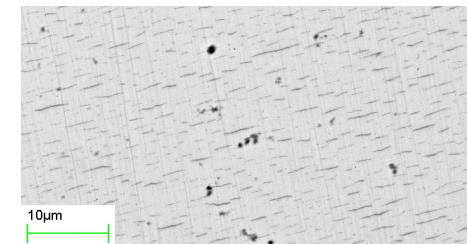
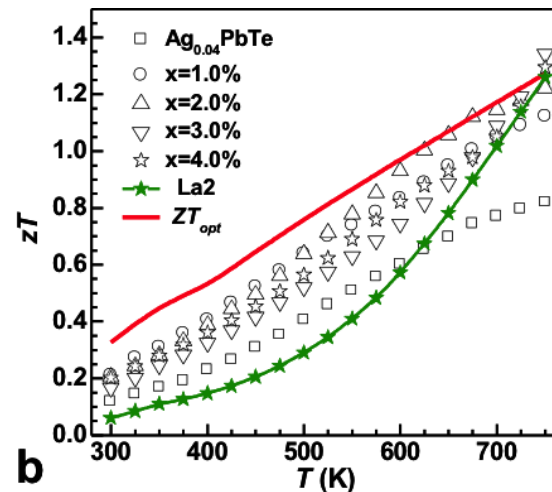
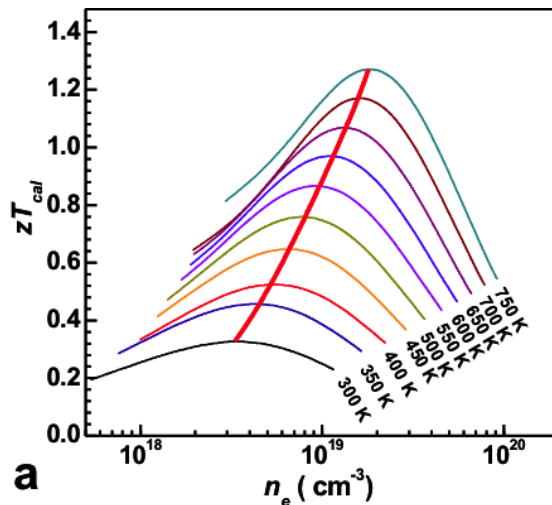
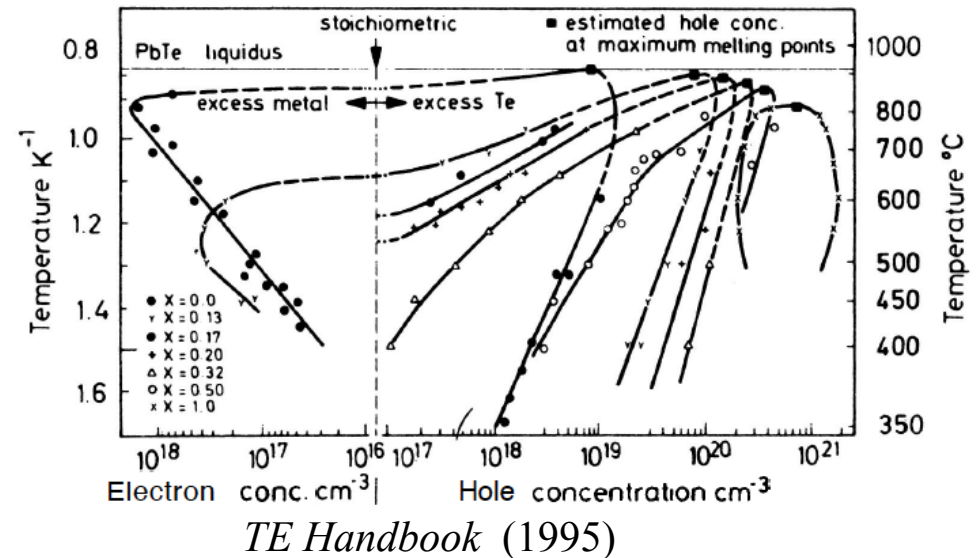


Temperature dependent
dopant solubility

Pb, Te, Ag, Cu in PbTe

B, P in SiGe

At solubility limit, n
increases with T



Ag, Ag₂Te particles
in PbTe

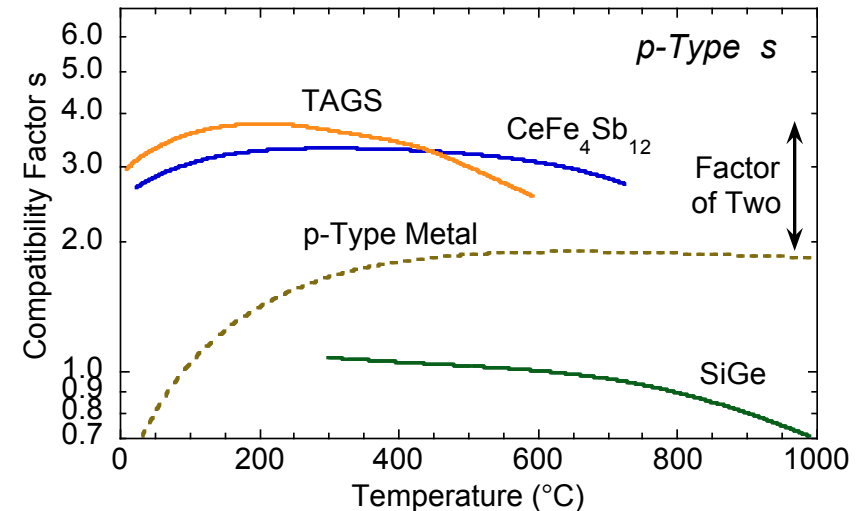
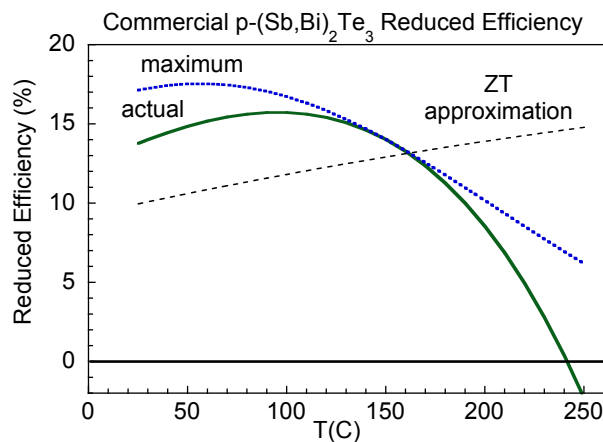
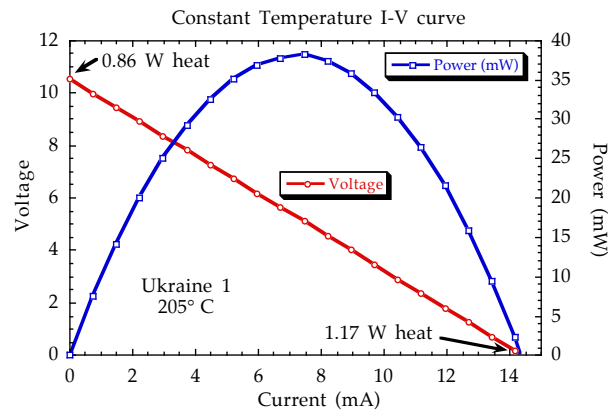


Compatibility Factor

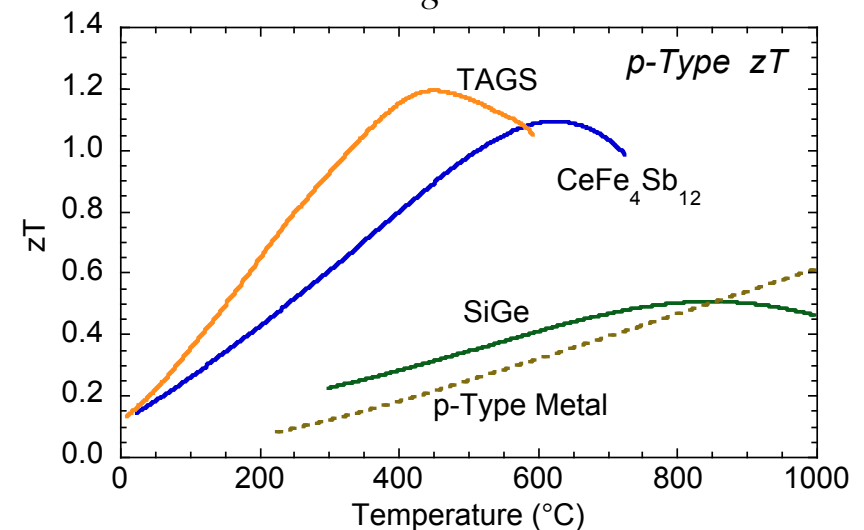
High Efficiency from zT
only with compatible s

$$s = \frac{\sqrt{1 + zT} - 1}{\alpha T}$$

operating current must match



*SiGe has small s compared to other TE
cant be segmented*



Thermoelectrics
Caltech Materials Science

Snyder *Applied Physics Letters* **84**, 2436 (2004)



Summary

Many levels of complexity for ZT

higher order effects relevant for high efficiency applications

Peak zT

carrier concentration for peak zT
quality factor maximization

$$B \sim \frac{N_V}{m_b^* K_L}$$

High average zT

carrier conc. tuning
functionally grading
*self-tuning composites

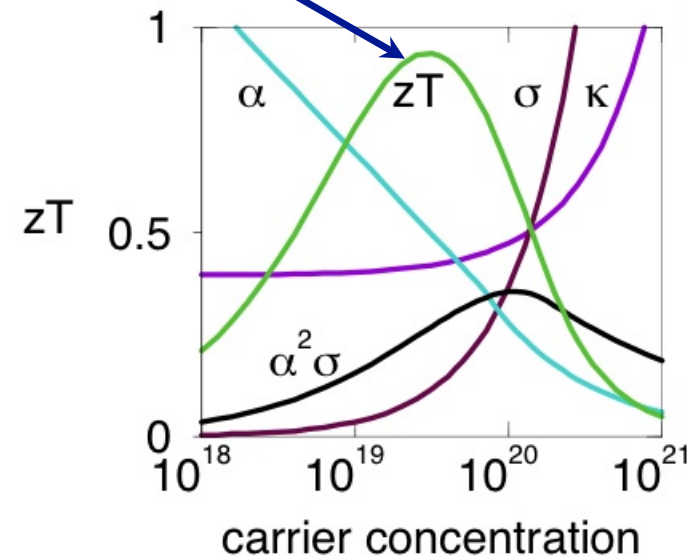
Band Engr. for ZT

quality factor engr for avg zT
nanostructures for low temp
stabilizing n^

Compatibility effects

ZT not a simple average of zT

Optimized carrier concentration
 $n \sim N_v^{2/3} m_b^* T^{3/2}$



Pei, Snyder, et al. *Advanced Materials* **23**, 5674 (2011)

Pei, Snyder, et al. *Advanced Energy Materials* **1**, 291 (2011)

Lalonde, Pei, Wang, Snyder, *Materials Today* **14** 526 (2011)



Thermoelectrics
Caltech Materials Science

Acknowledgements



Postdocs

Yanzhong Pei

Aaron Lalonde

Shiho Iwanaga

Eric Toberer



Yangzhong PEI



Aaron Lalonde



Shiho IWANAGA



Eric Toberer

Students

Heng Wang

Alex Zevalkink

Nick Heinz

Greg Pomrhen

Andrew May



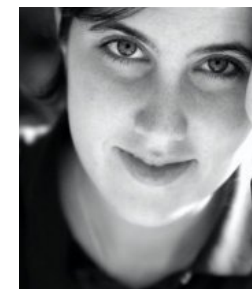
Heng WANG



Greg Pomrhen



Nick Heinz



Alex Zevalkink

Collaborations/discussions

Lidong Chen

Xiaoya Shi (SIC-CAS)

Norb Elsner

Joseph Heremans

Mercuri Kanatzidis

Yaniv Gelbstein

David Singh (ORNL)

Jean-Pierre Fleurial

Sabah Bux (JPL)

Axel van de Walle (Caltech)

Funded by

NASA-JPL

Beckman Foundation, Caltech

DARPA

Sandia National Laboratories

AFOSR MURI



Thermoelectrics
Caltech Materials Science